

Grazing Incidence Small Angle X-ray Scattering

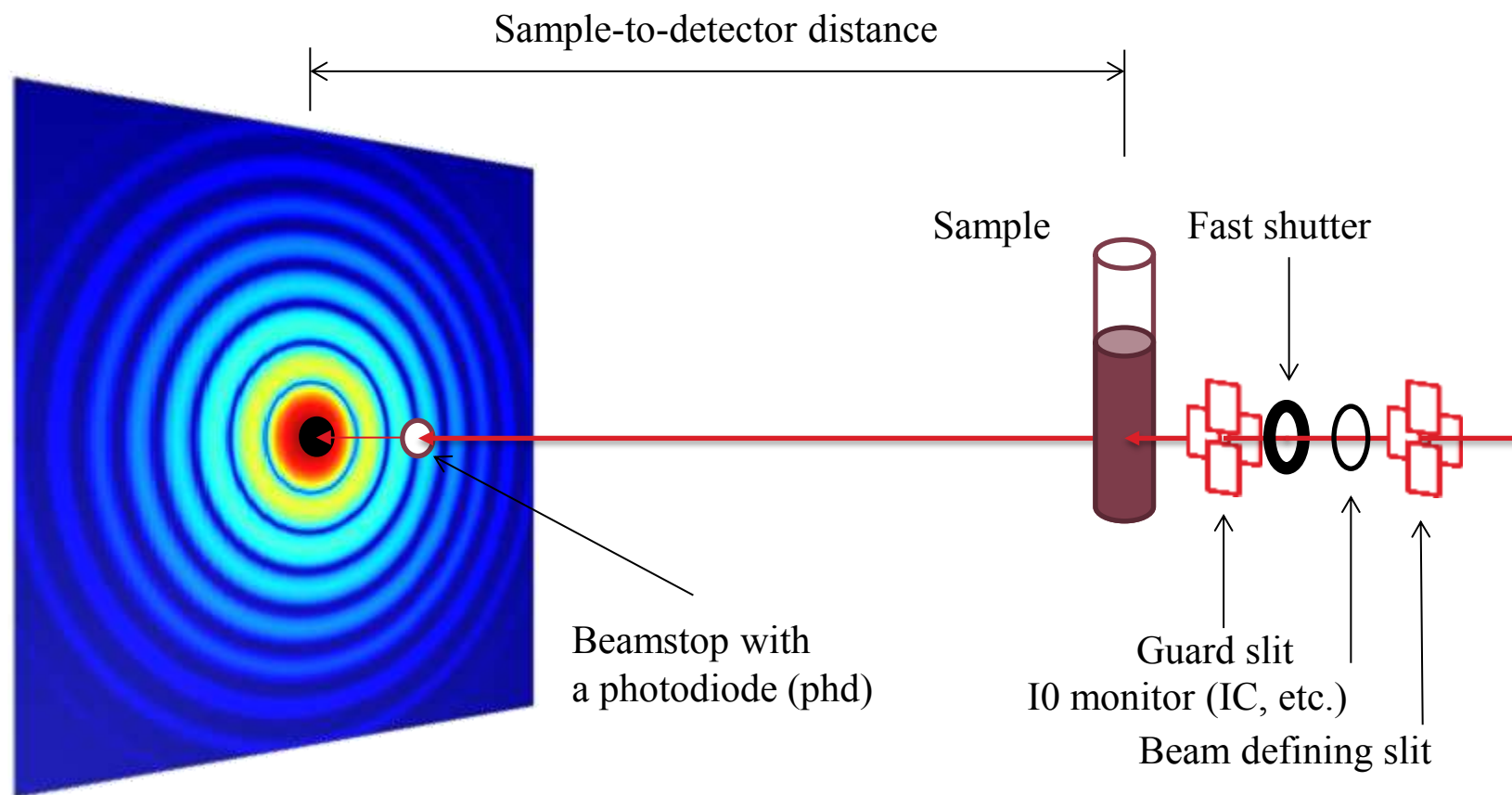
Byeongdu Lee

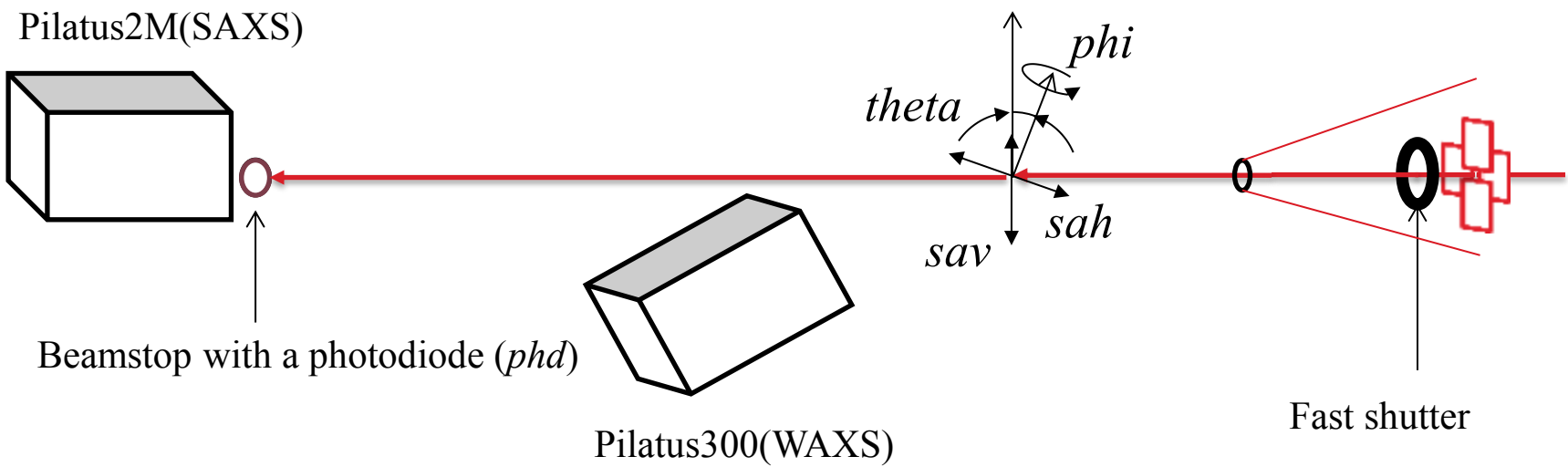
X-ray Science Division/Advanced Photon Source

Outline

- Grazing Incidence Small Angle X-ray **Scattering** (GISAXS) beamline and setup
- Key knowledge to interpret GISAXS
 - Fourier Transform
 - Shape, size, and orientation of particles and lattices
- The effect of a small incident angle
 - Reflection
 - Experimental examples: Crystalline nano particles.
 - Penetration depth
 - Experimental examples: Block copolymer films
 - Refraction
- Grazing Incidence Small Angle X-ray **Diffraction**
 - Ewald sphere
 - Experimental examples: Block copolymer films
- Quantitative calculation
 - Distorted wave Born approximation
 - Snell's law and Fresnel's law





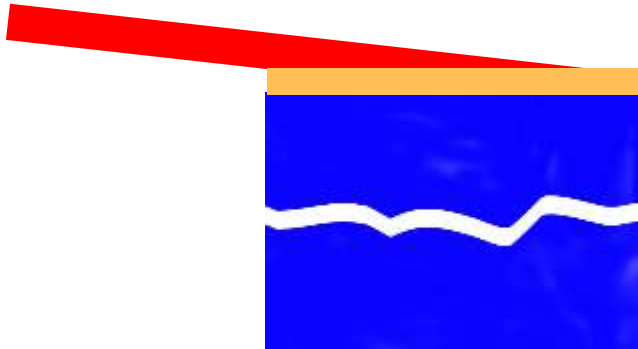


12ID-B beamline at APS

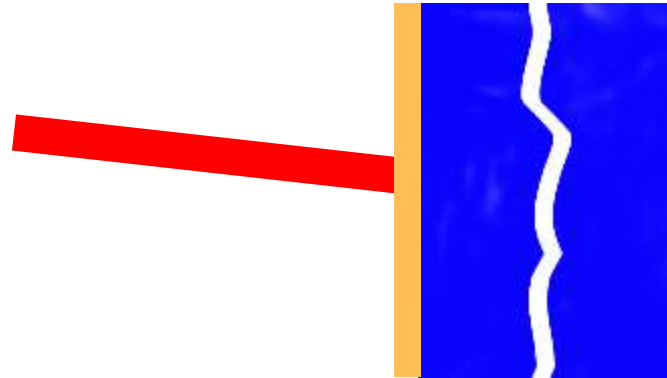


Why GISAXS?

GISAS



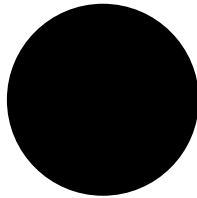
SAS



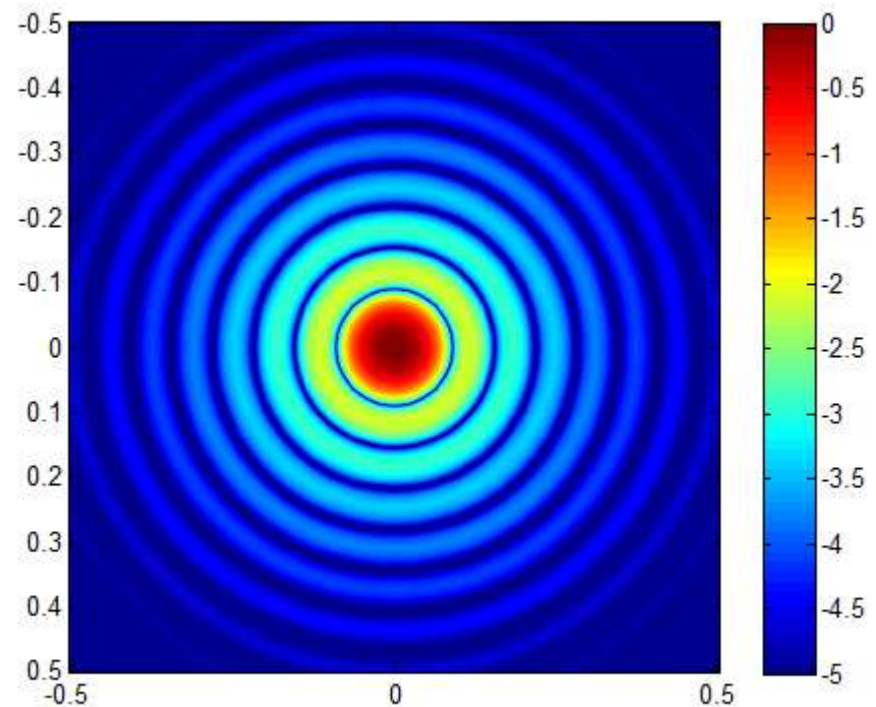
Advantage for the grazing incidence geometry for thin film.

- 1. Several orders larger scattering volume*
- 2. Scatterings from oriented samples*

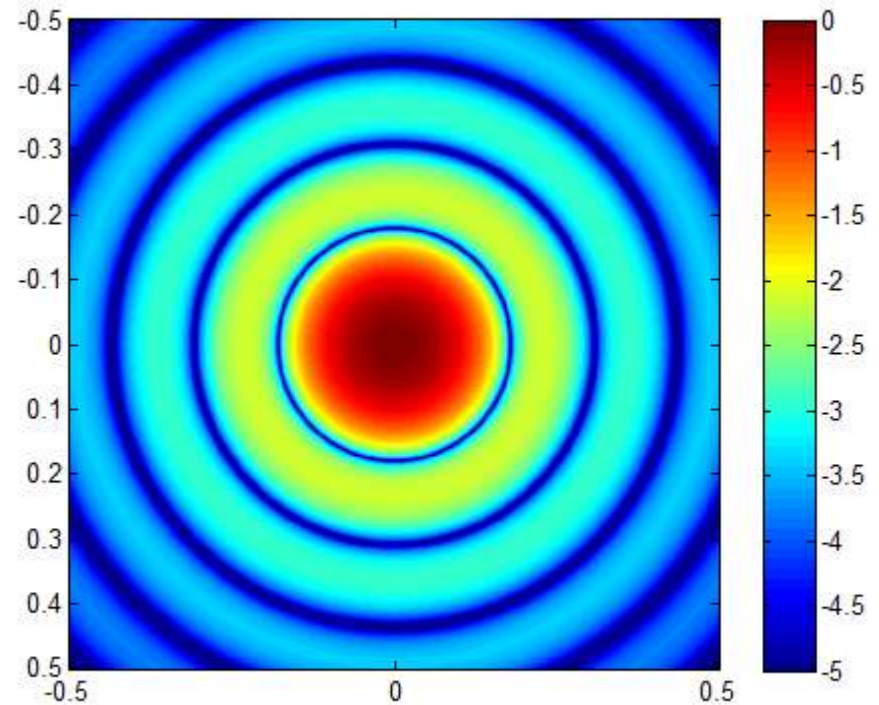
Scattering and Fourier Transform



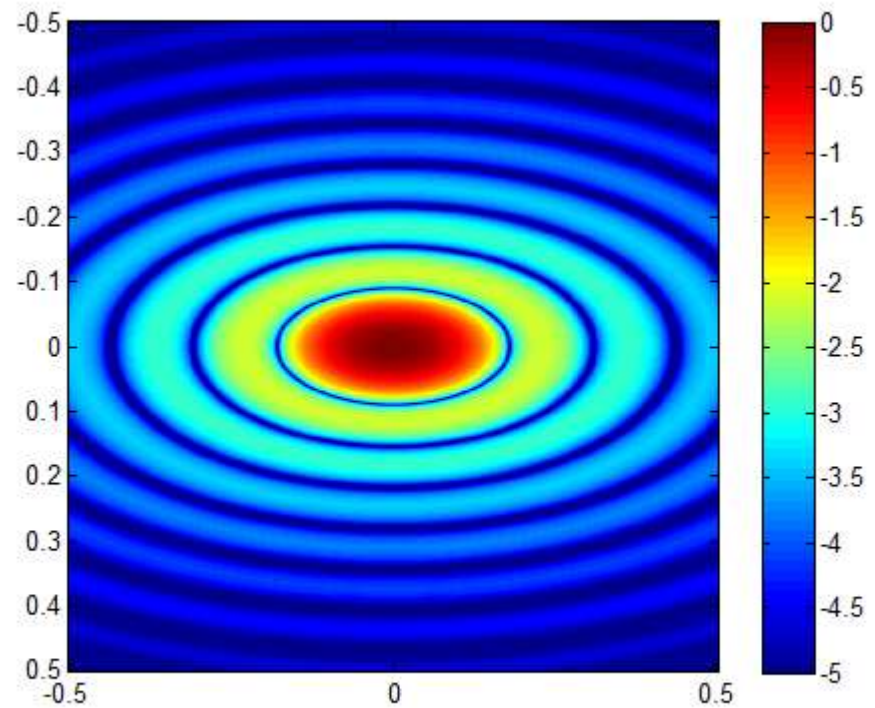
Electron density



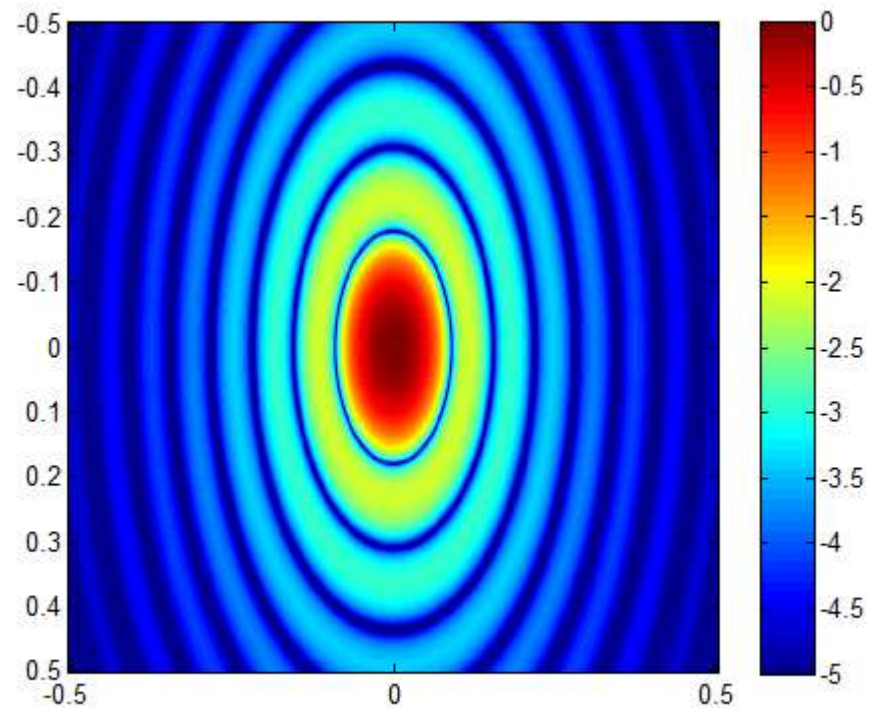
$|\text{FT}(\text{Electron density})|^2$

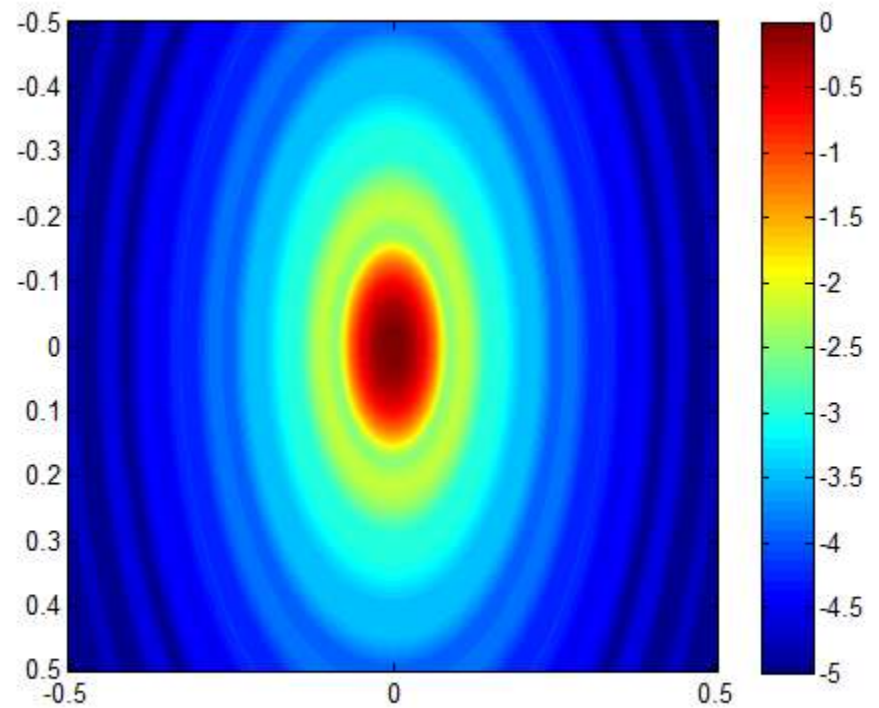
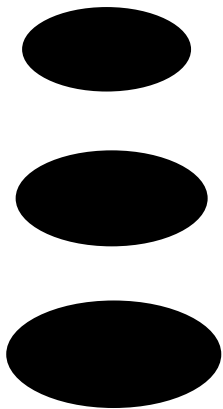


The Bragg equation: $q = 2\pi/D$

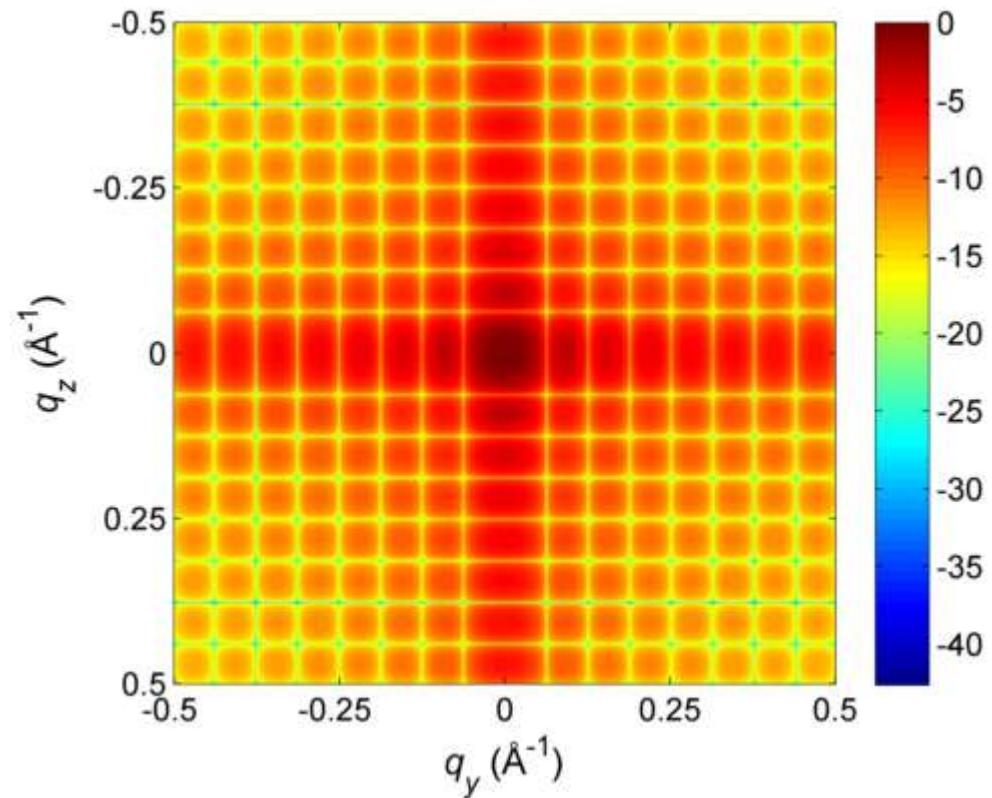
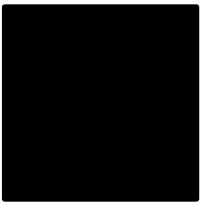


1. The Bragg equation: $q = 2\pi/D$
2. Orientation

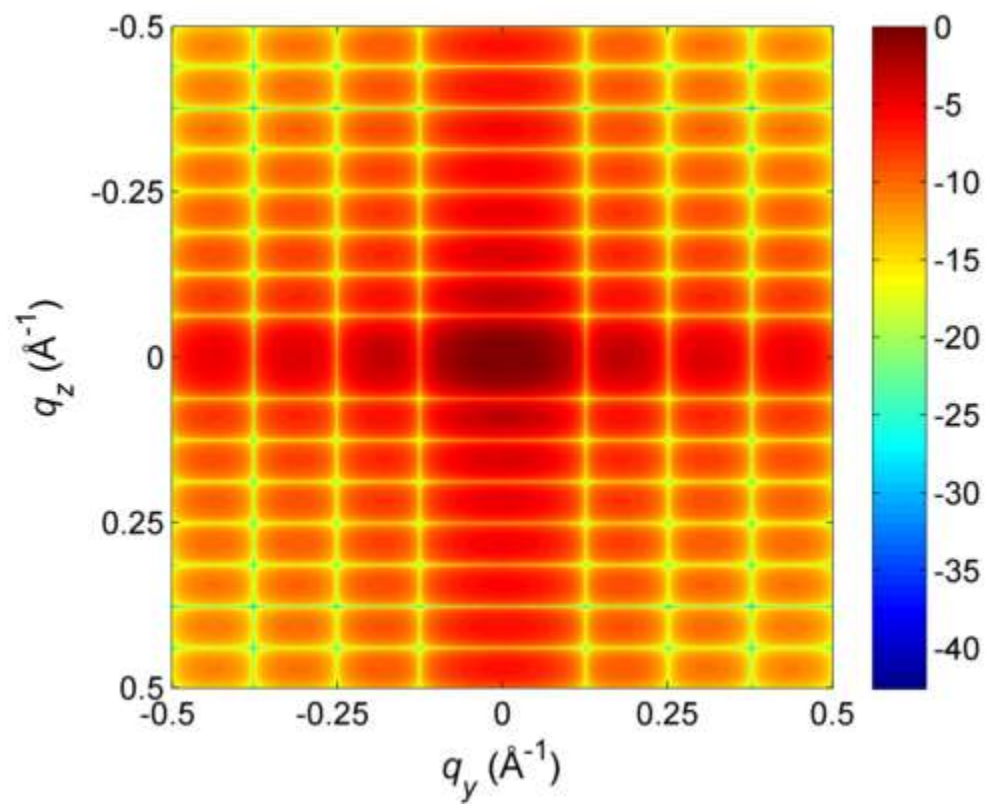


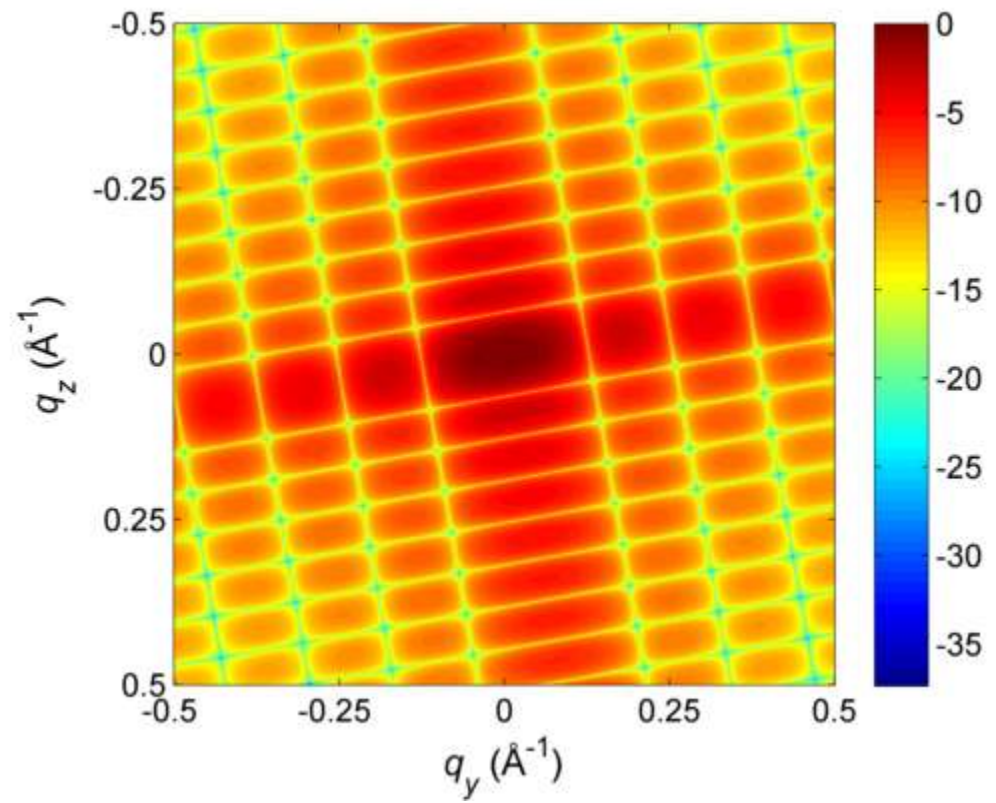
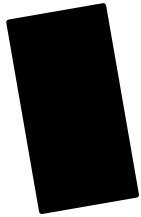


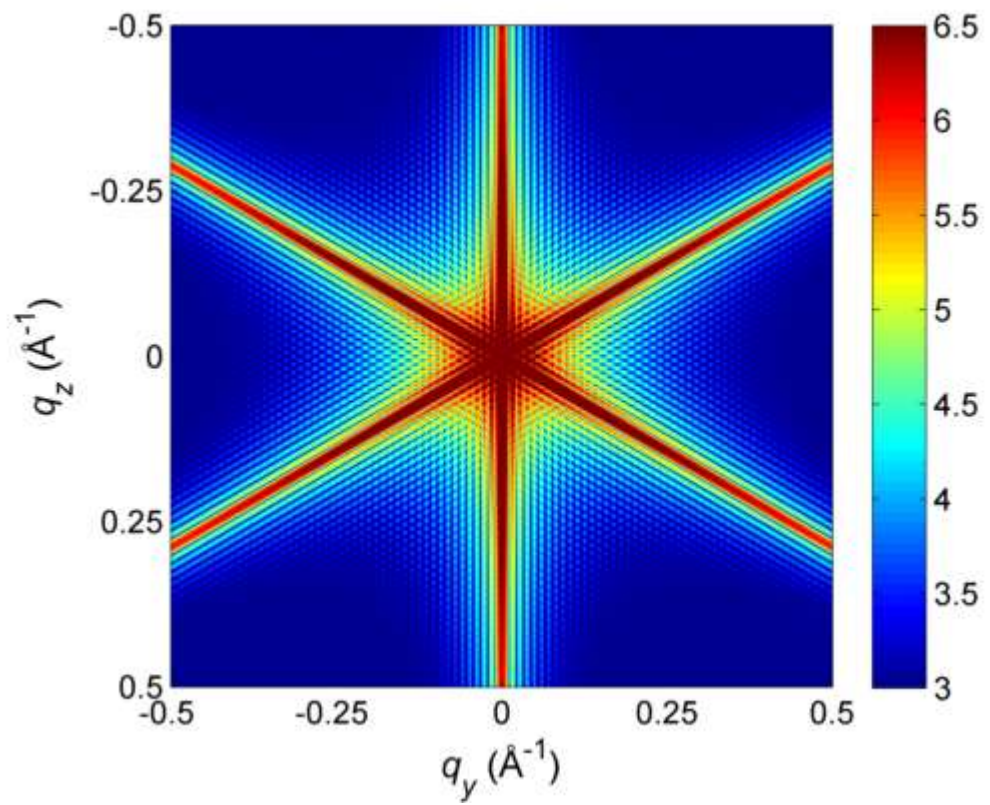
Polydispersity

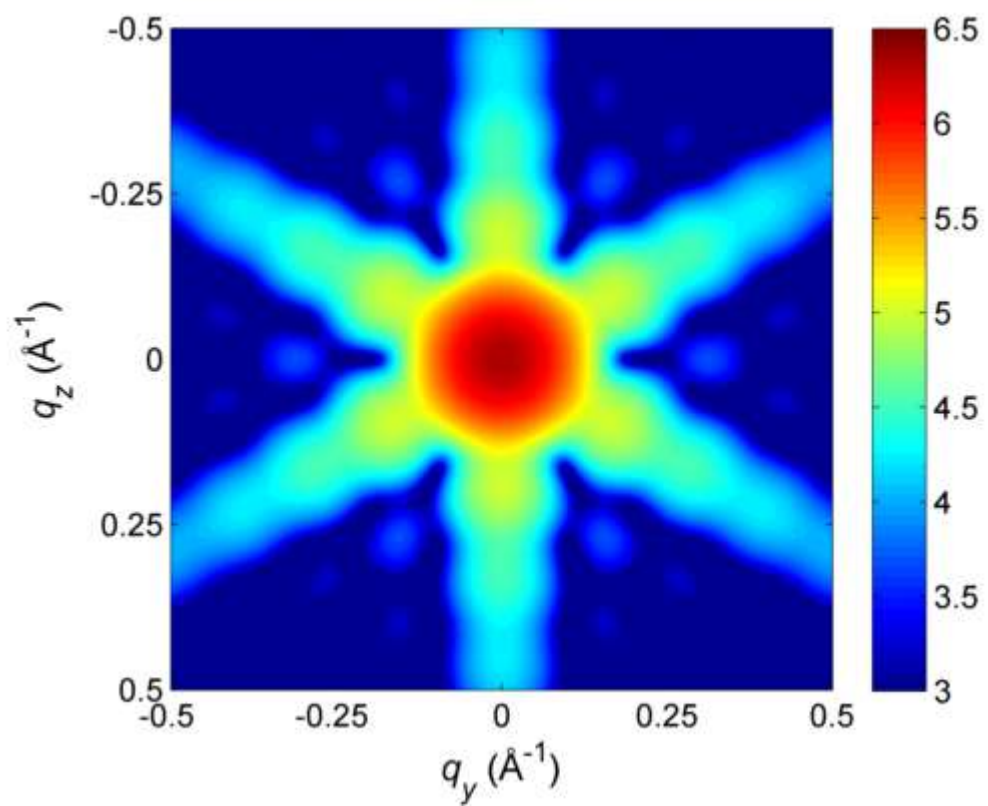


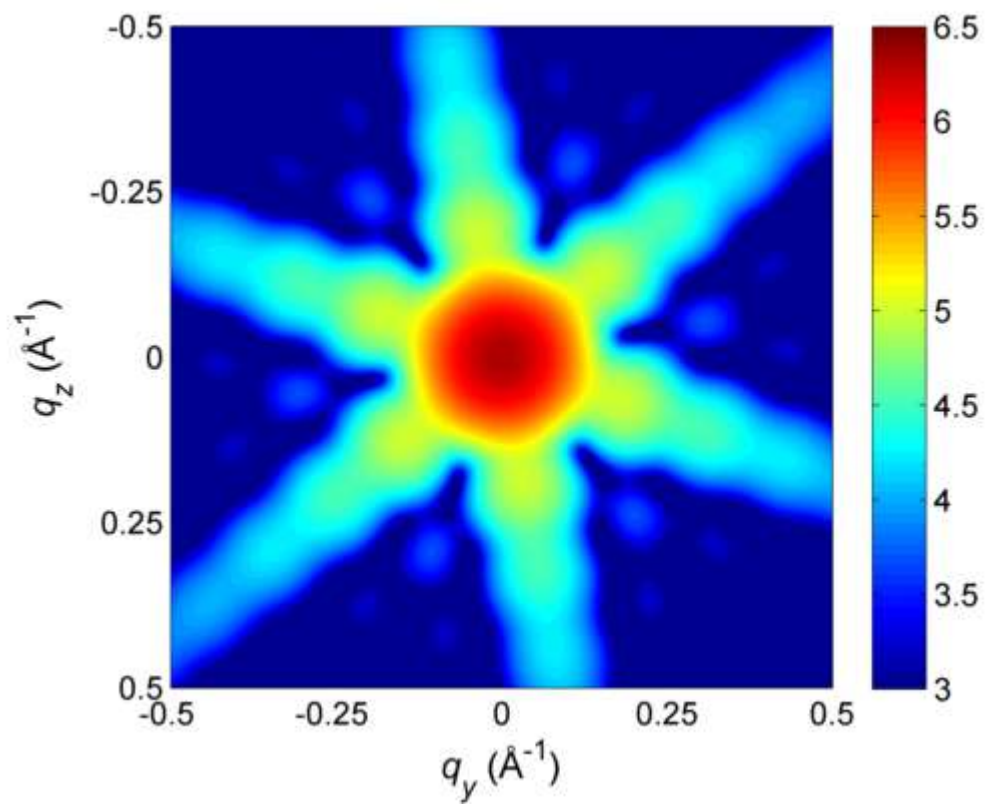
1. Scattering from a flat surface
2. Particle shape



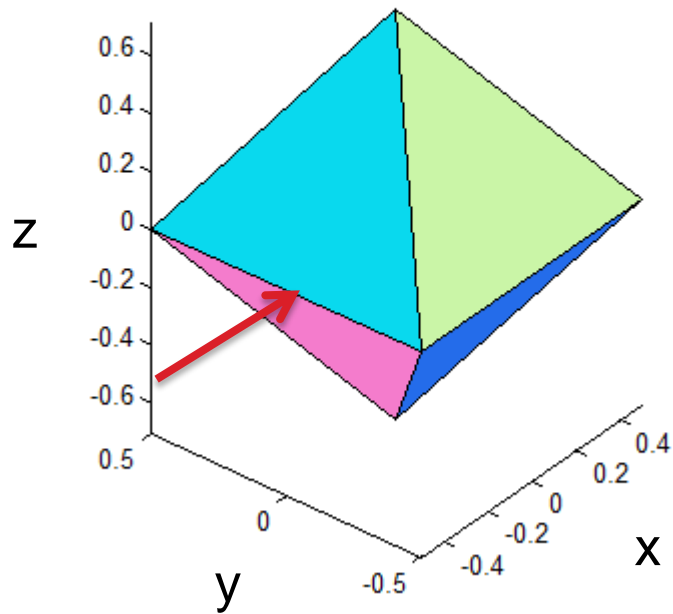




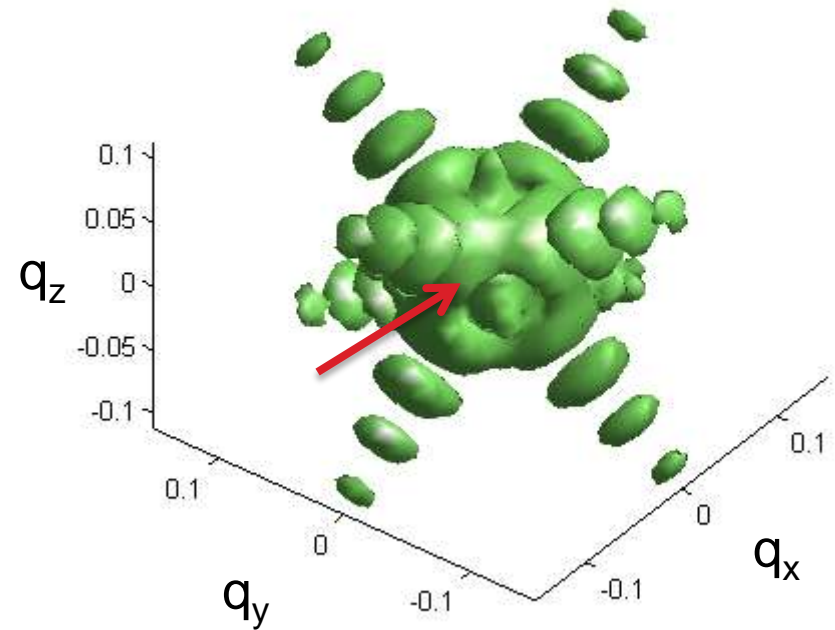




Octahedron

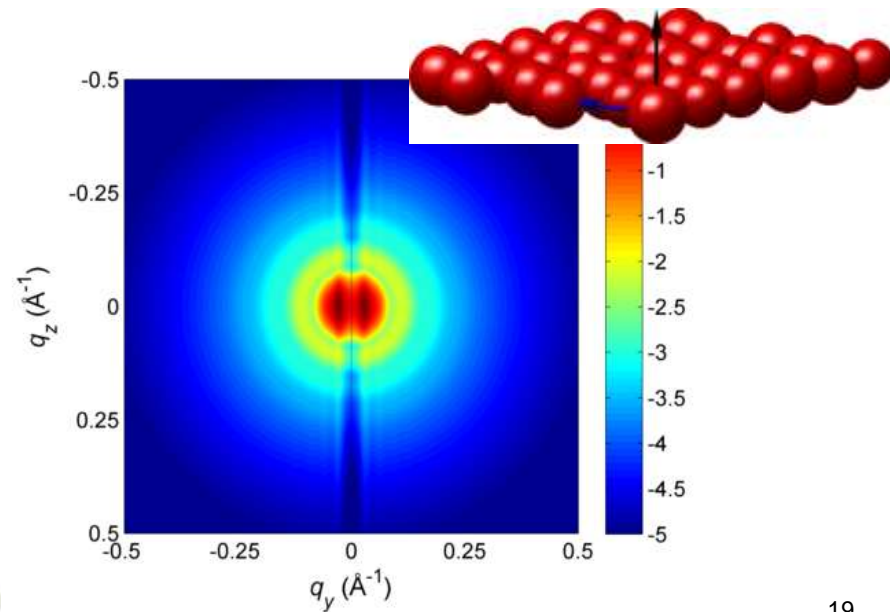
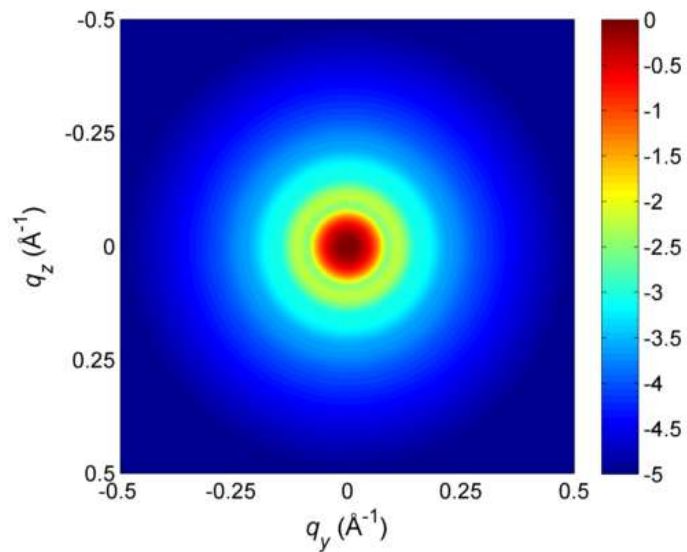
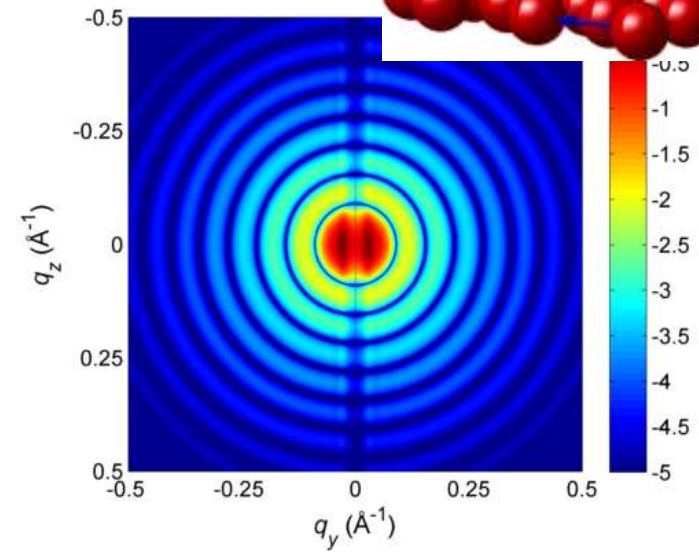
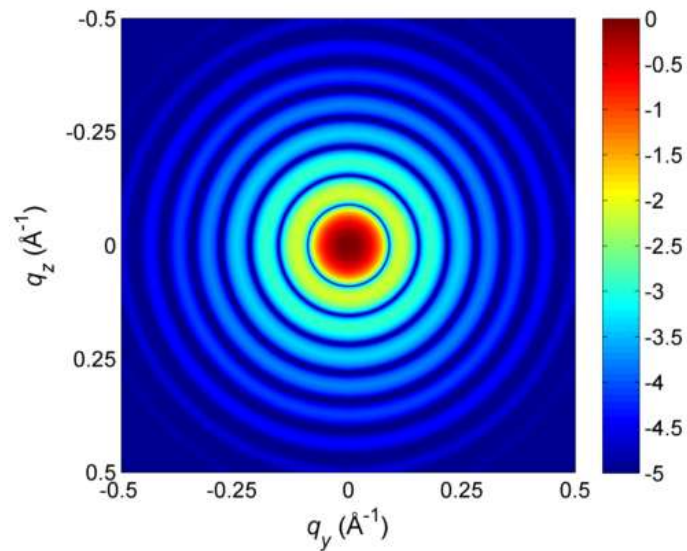


Real space

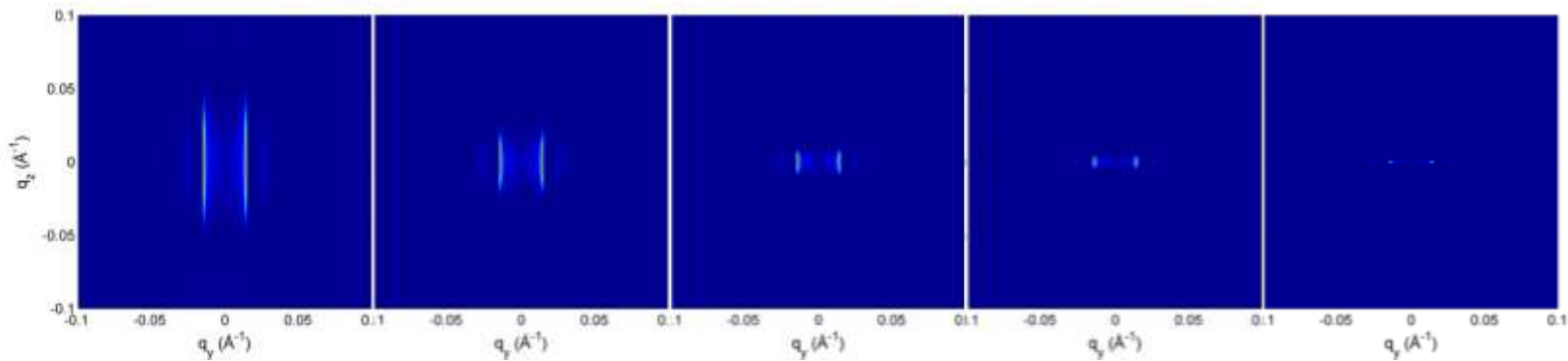
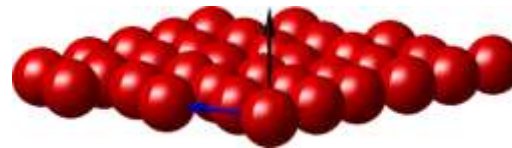
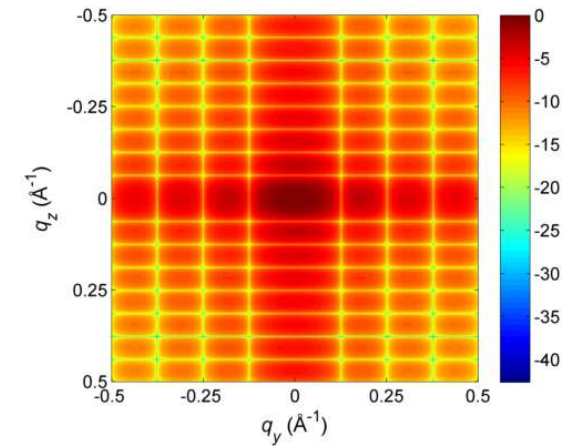


Reciprocal space

When the structure factor is applied



2D cylinders



Height 10nm

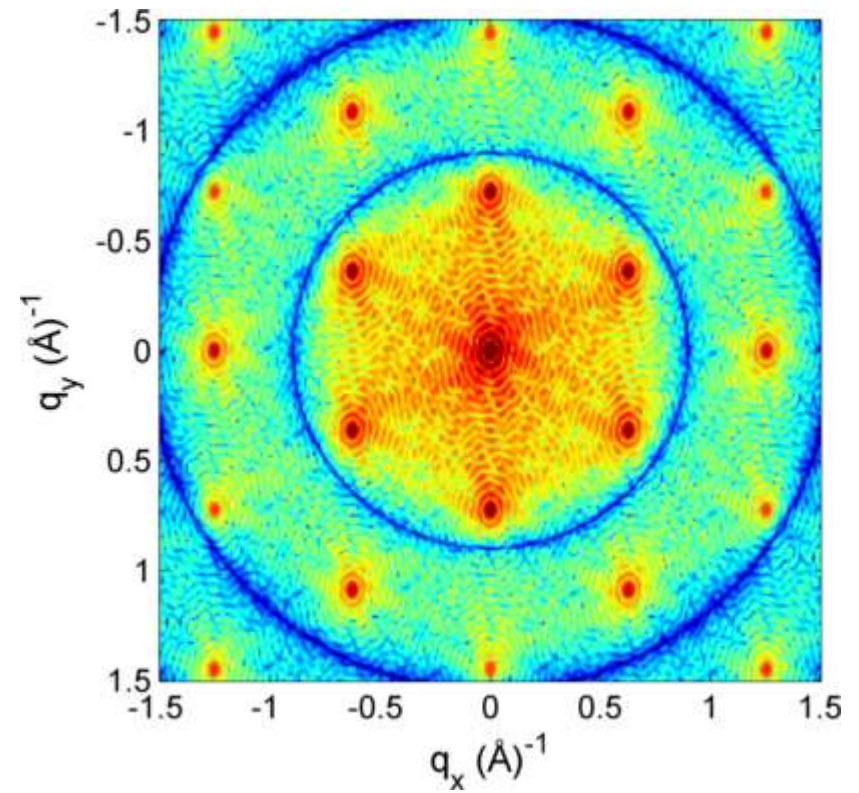
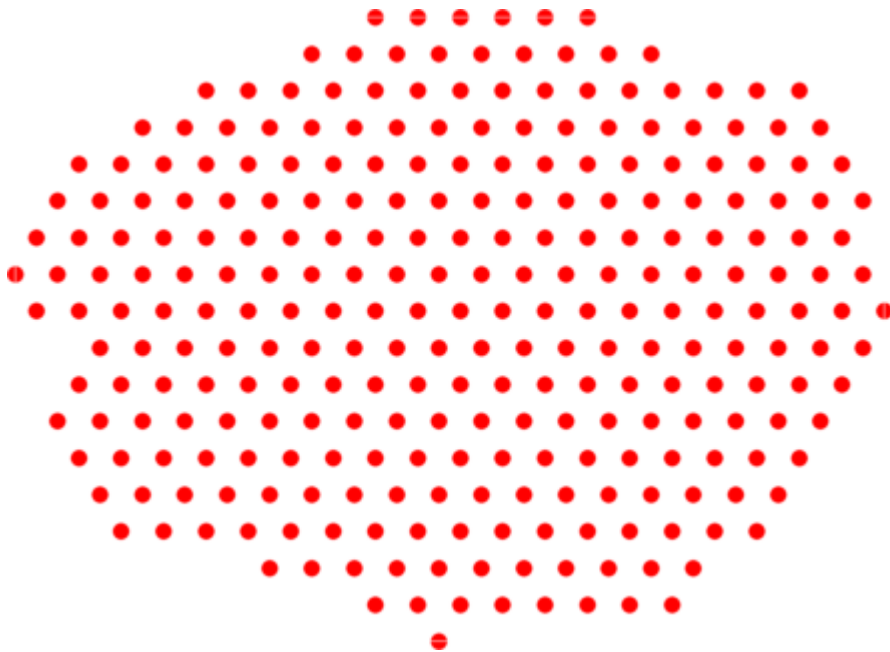
20nm

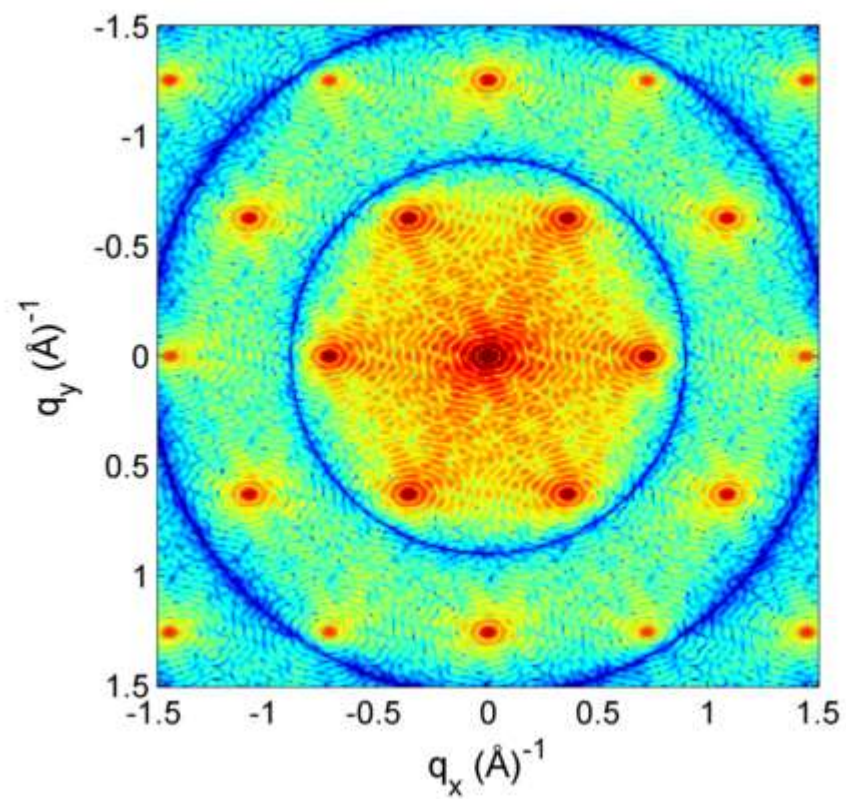
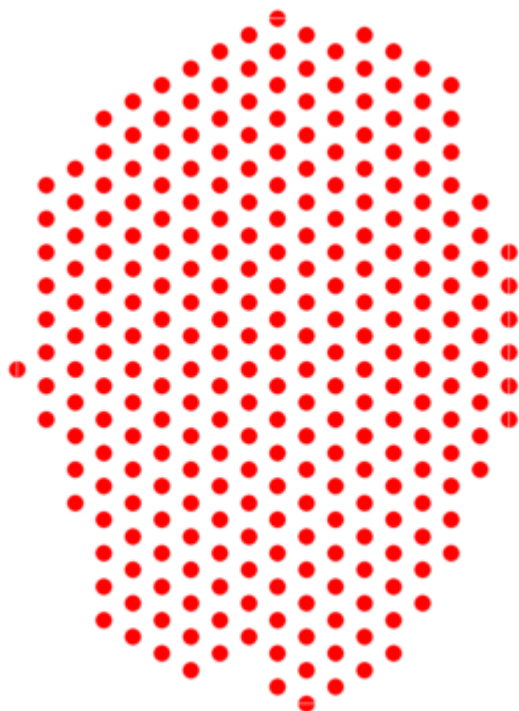
50nm

100nm

500nm

lattices





Note

- So far, the intensities have been calculated
NUMERICALLY
 - $I(\mathbf{q}) = \left| \sum_{i=1}^N F_i(\mathbf{q}) \right|^2$
 - Takes a long time to model and calculate.
- Analytical ways will follow

How to split the form factor and the structure factor

- $I(\mathbf{q}) = \left| \sum_{i=1}^N F_i(\mathbf{q}) \right|^2 = P(\mathbf{q})S(\mathbf{q})$
- Decoupling approximation (DA)
- Local Monodisperse Approximation (LMA)



The form factor

$$F(\mathbf{q}) = \rho \int_V e^{-j\mathbf{q}\cdot\mathbf{r}} d\mathbf{r}$$

$$\begin{aligned} P(q) &= \langle \overline{|F(q)|^2} \rangle \\ &= \sum_{l=1}^{N_p} \frac{N_l}{N_p} \langle |F_l(q)|^2 \rangle \\ &= \int n(r) \langle |F_l(q)|^2 \rangle dr \end{aligned}$$

The form factor models from Babonneau's software

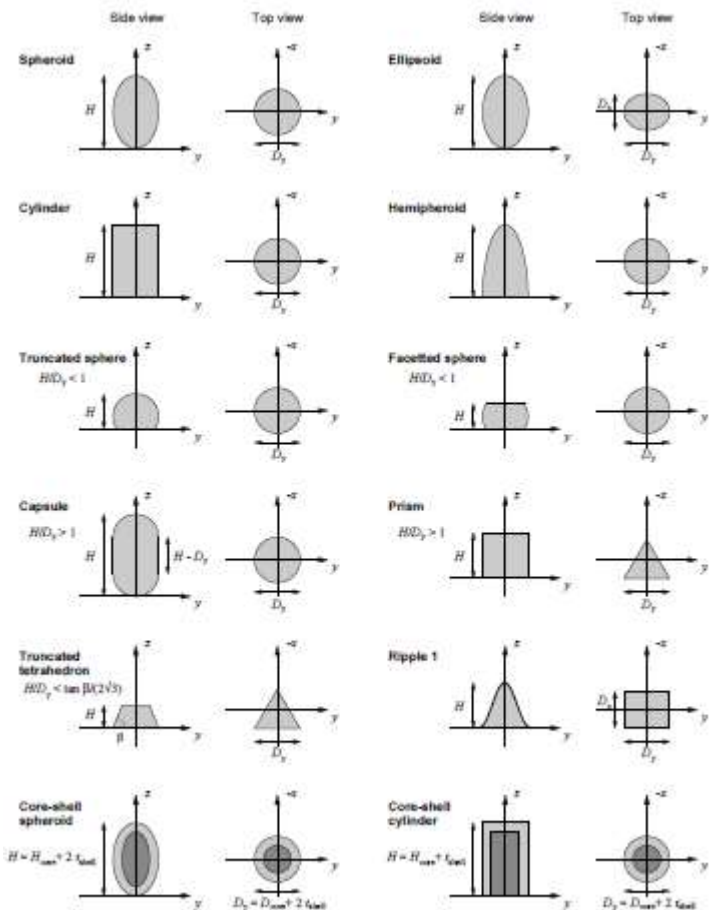


FIGURE 3 – Form factors included in the FitGISAXS package.

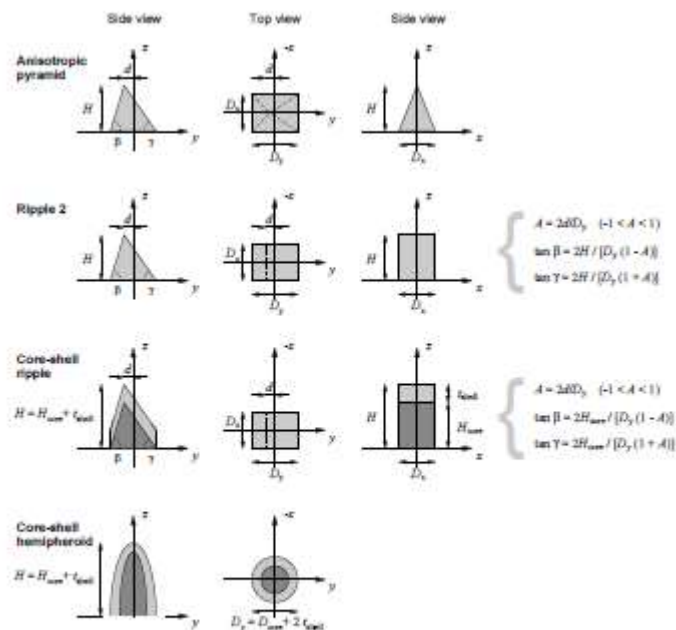
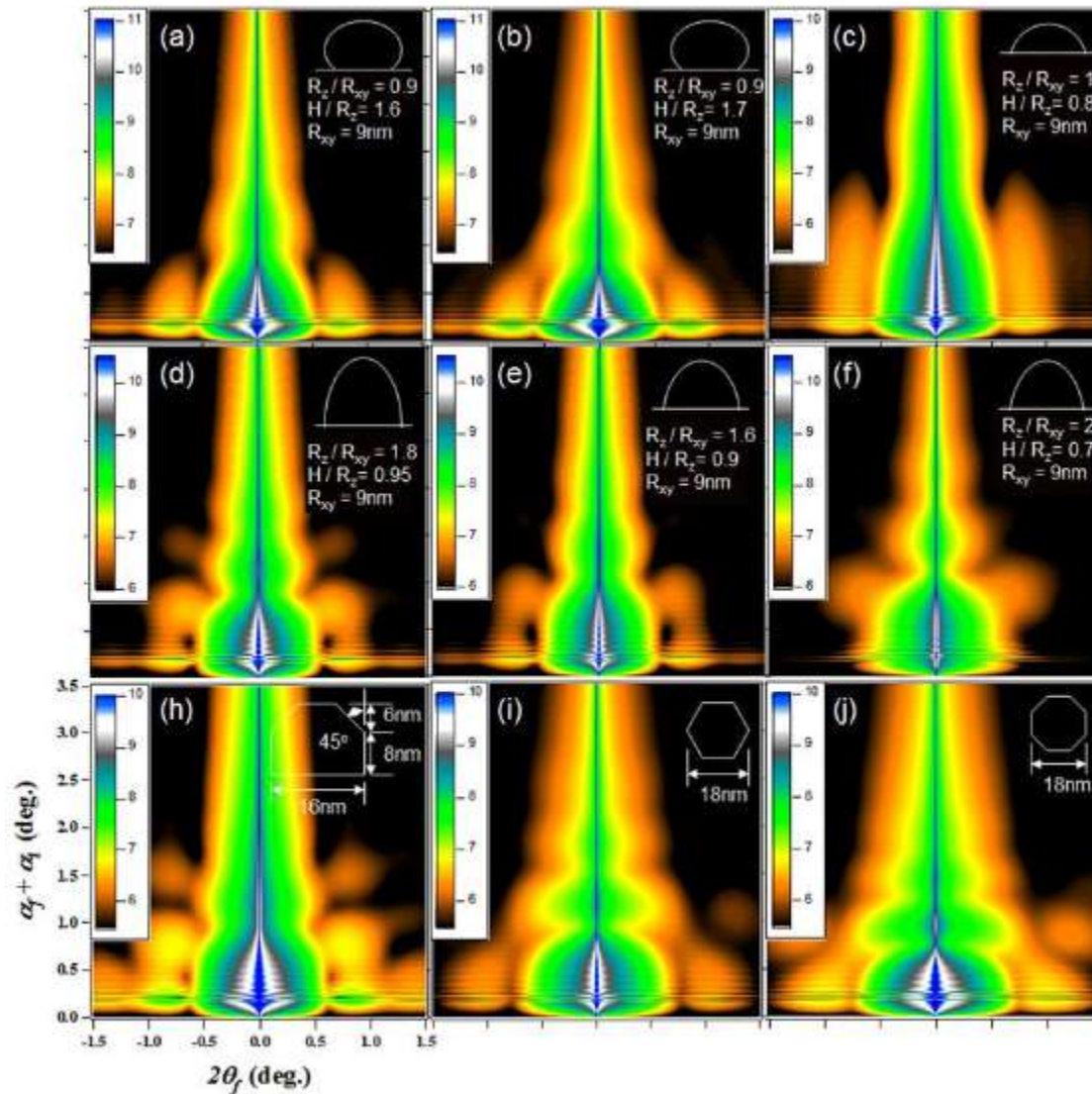


FIGURE 4 – Additional form factors included in the FitGISAXS package.

Rods laid down on a substrate



B. Lee et al. Langmuir, 2007, 23 (22), pp 11157–11163

The size distribution model

- Gaussian
- Double Gaussian for the bimodal distribution.
- Log-normal
- Double Log-normal for the bimodal distribution.
- Weibull
- Schultz-Zimm function



The structure factors from Babonneau's software

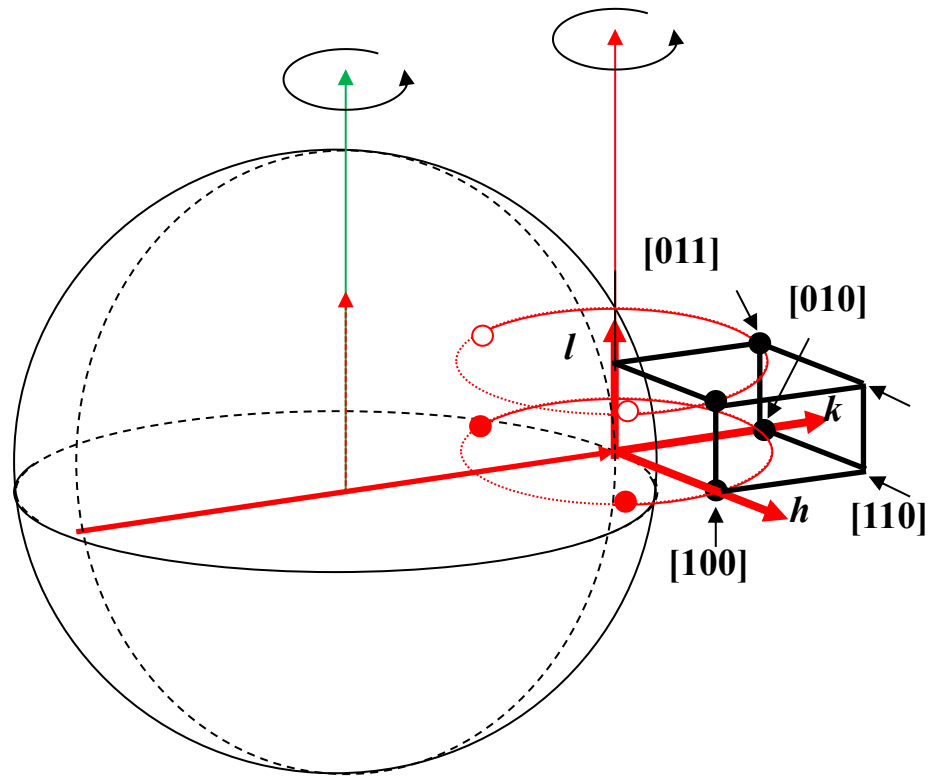
- Random organization: $S(q) = 1$
- Percus-Yevick 3D : Hard sphere potential
- Percus-Yevick 2D
- Paracrystal 1D
- Paracrystal 2D rectangular
- Paracrystal 2D hexagonal
- And many others.. See IsGISAXS manuals.

Reciprocal space or scattering vector q

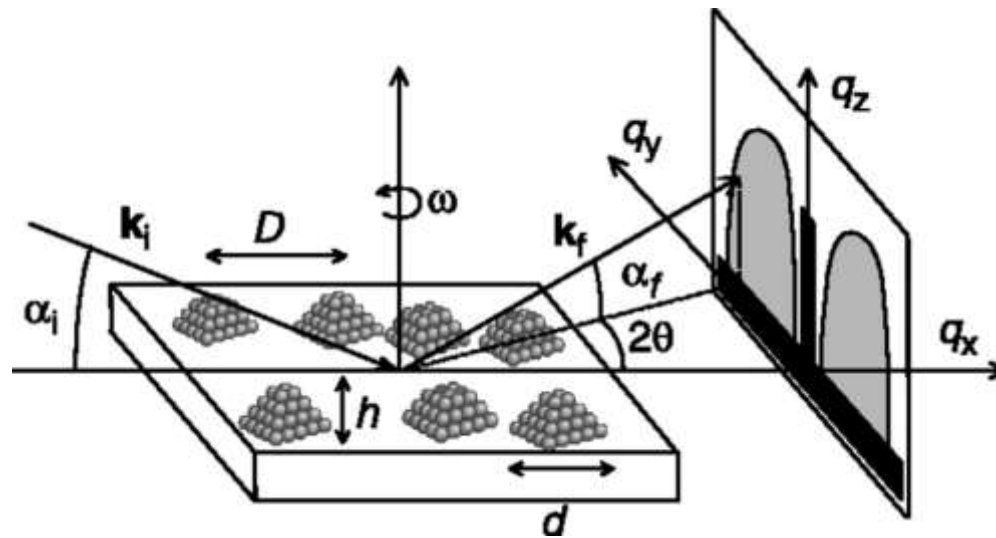
- The scattering vector q is not a scalar but a vector. (q_x, q_y, q_z)
- When a lattice (or a sample) rotate, the reciprocal lattice (or the scattering from the sample) rotates.
- The Bragg condition is not only $q = 2\pi/d$.



Ewald sphere



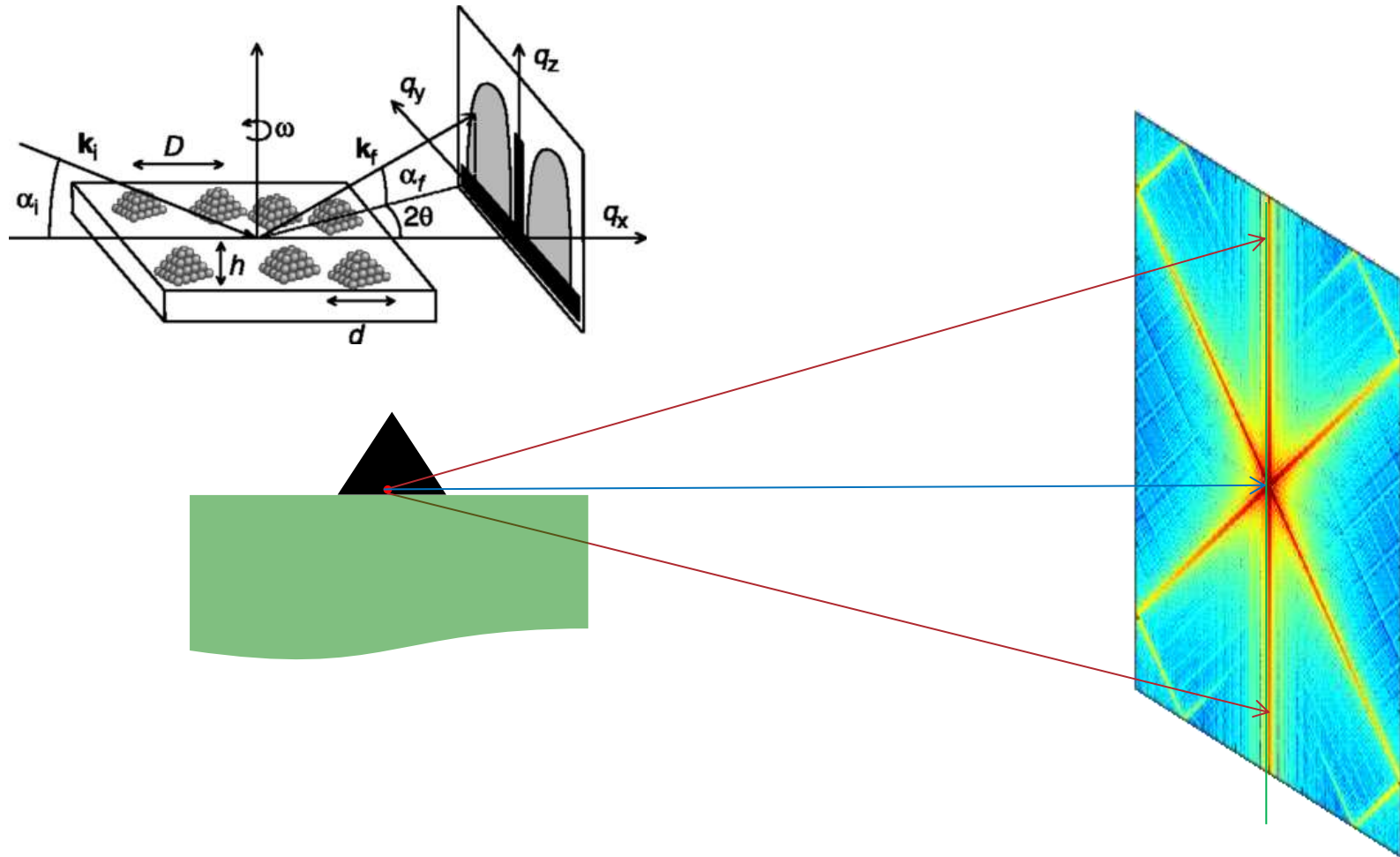
Definitions of angles and the q space



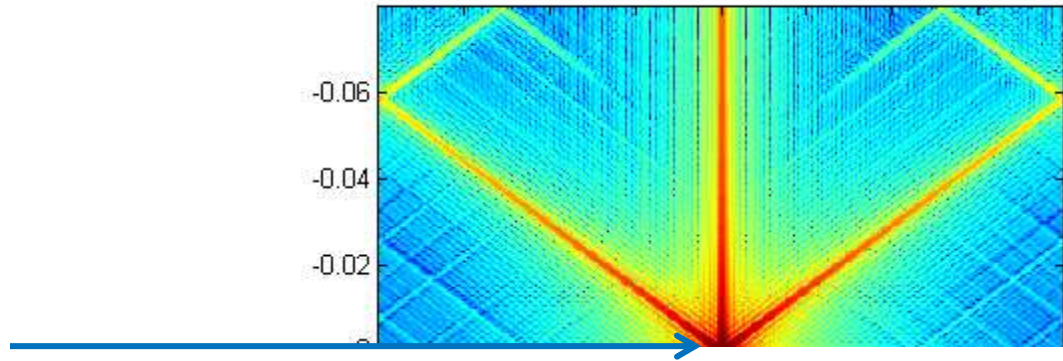
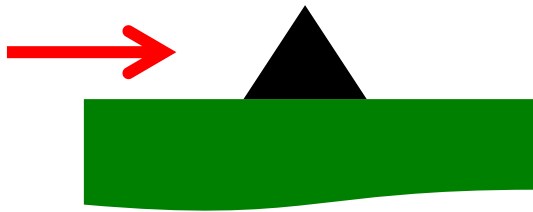
Note

- Effects due to the grazing incidence geometry
 - Absorption
 - Reflection
 - Refraction
- These effects are highly depending on the sample.
 - Supported island.
 - Buried particles.
 - Sandwiched particles.
 - ...

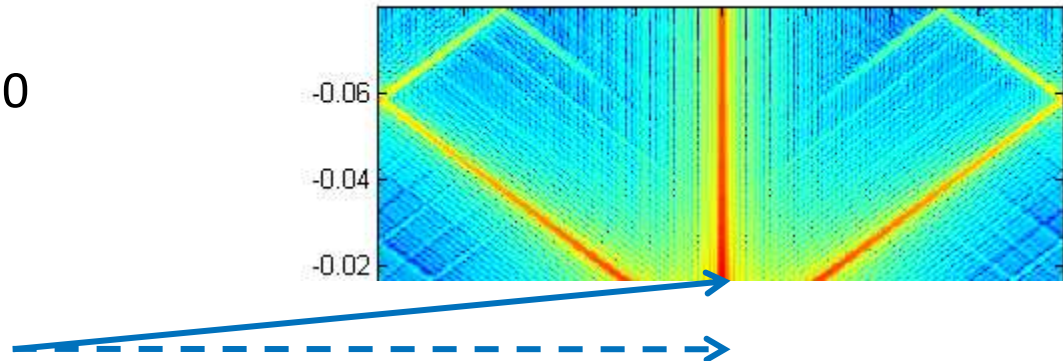
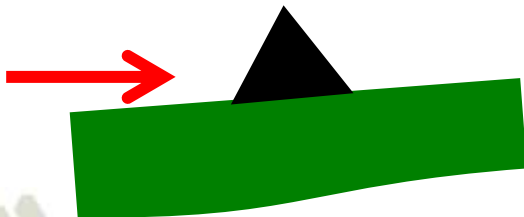
In GISAXS, you can measure only the upper half
- Substrate absorbs the downward scattering



At the incident angle, 0

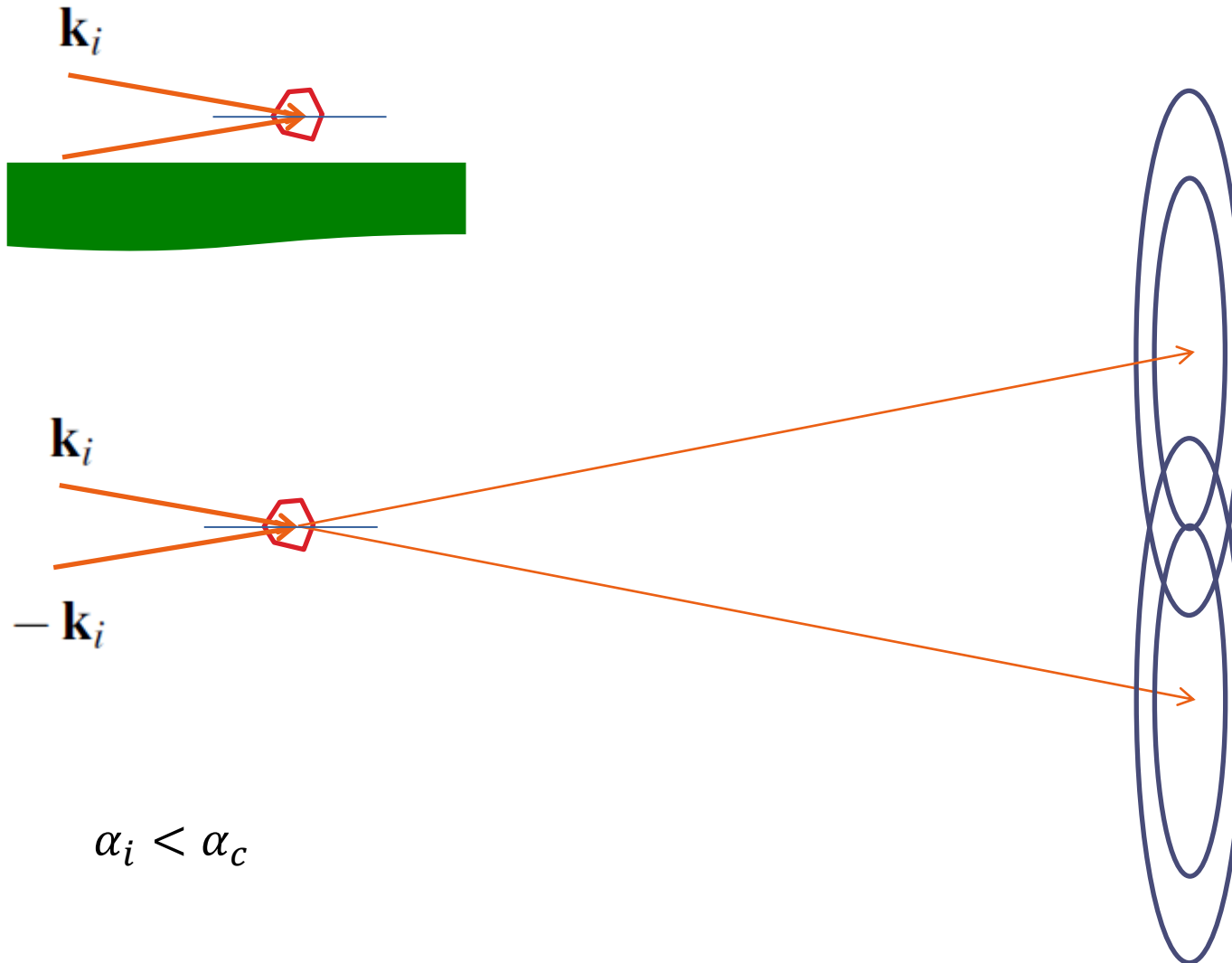


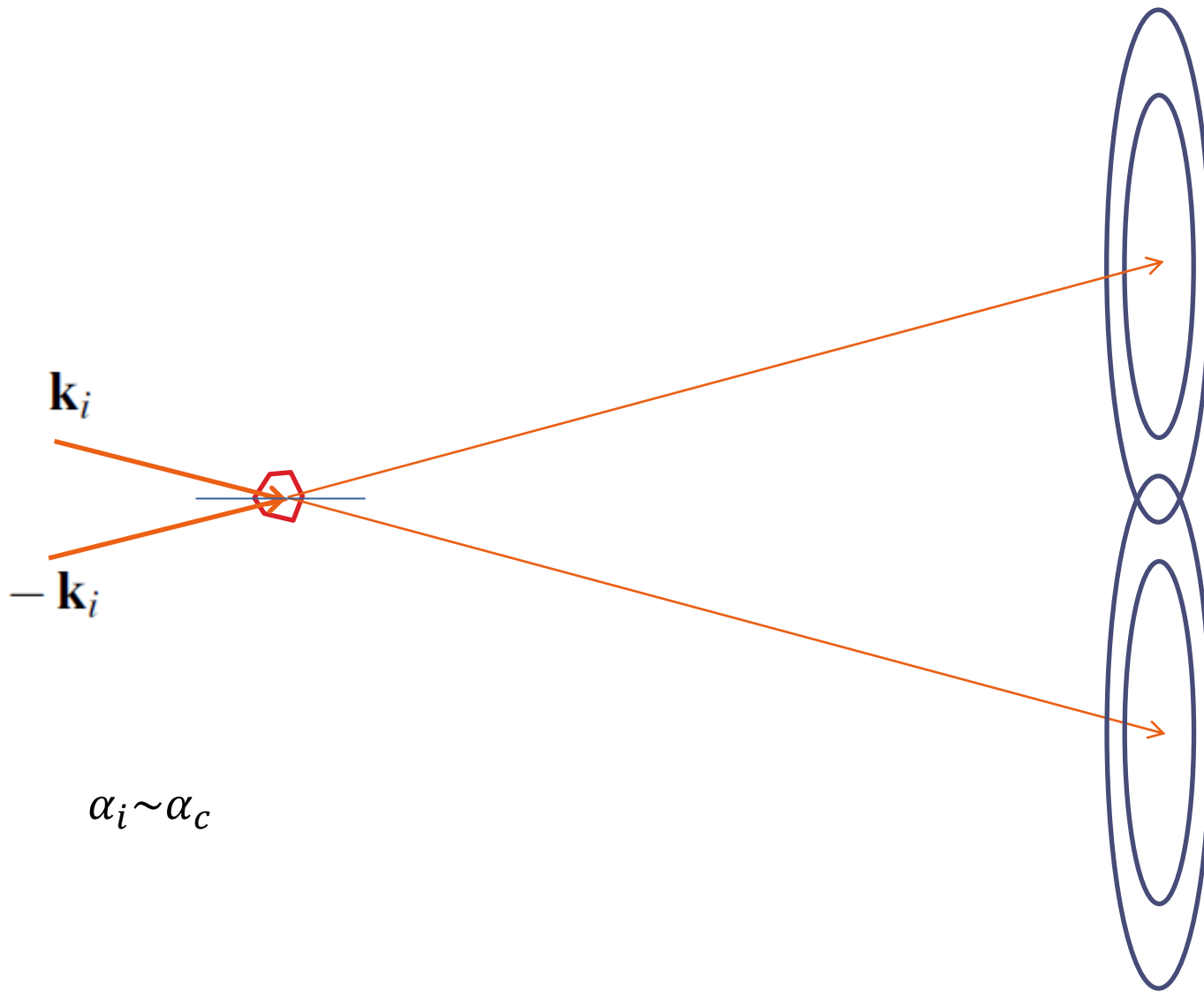
At the incident angle > 0

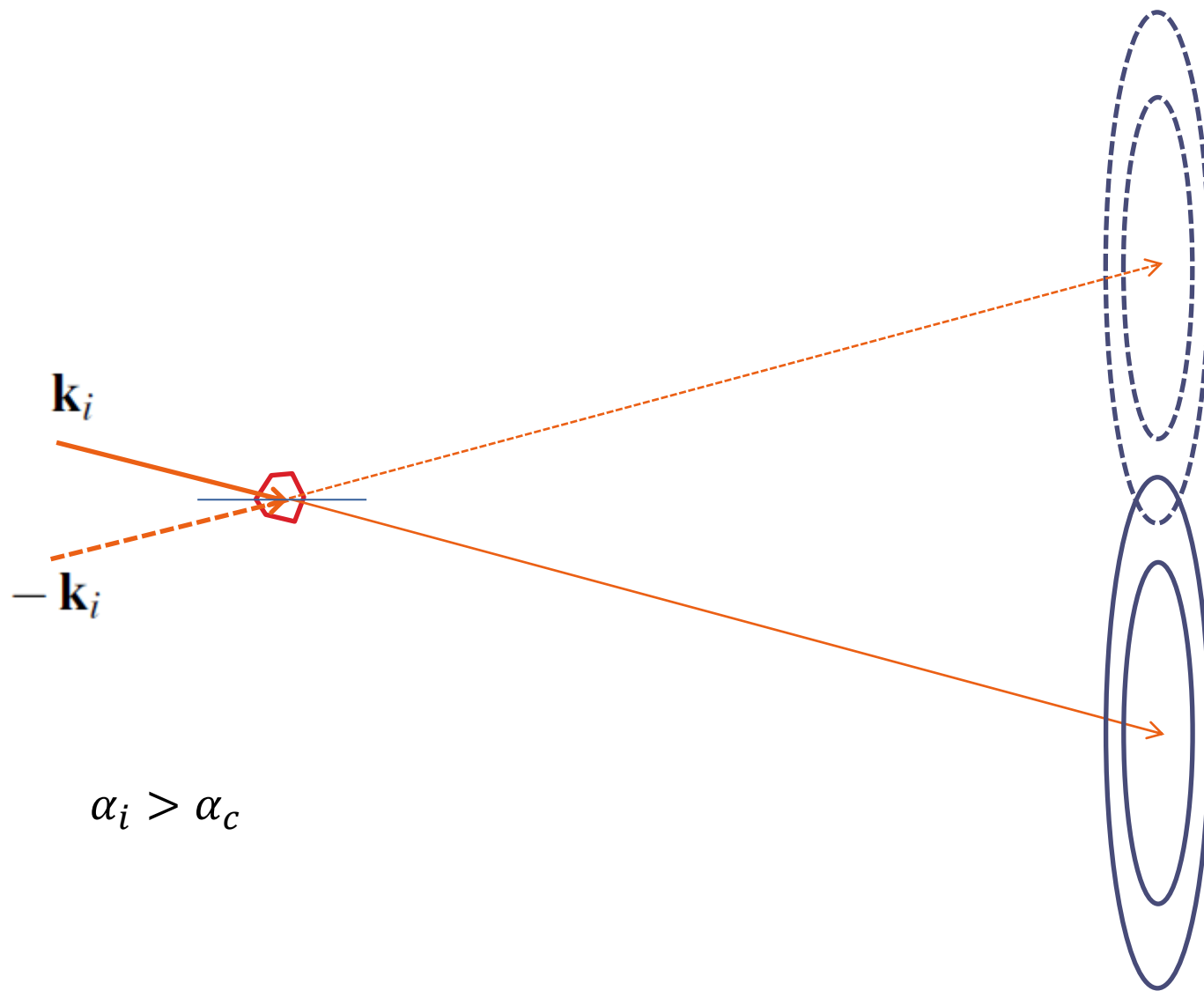


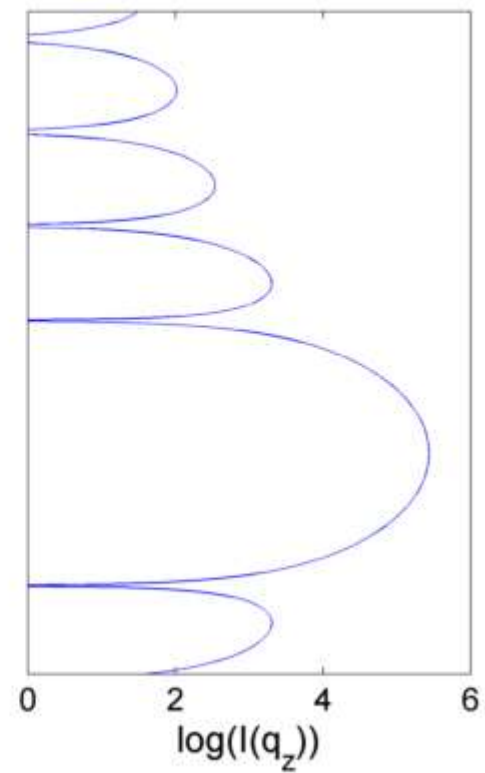
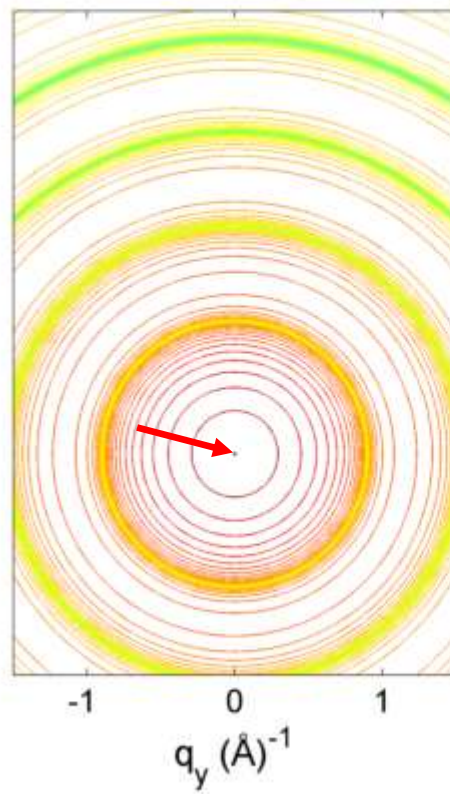
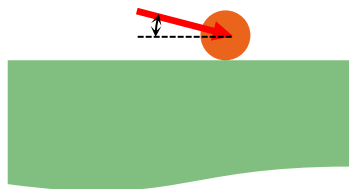
The effect of a small incident angle

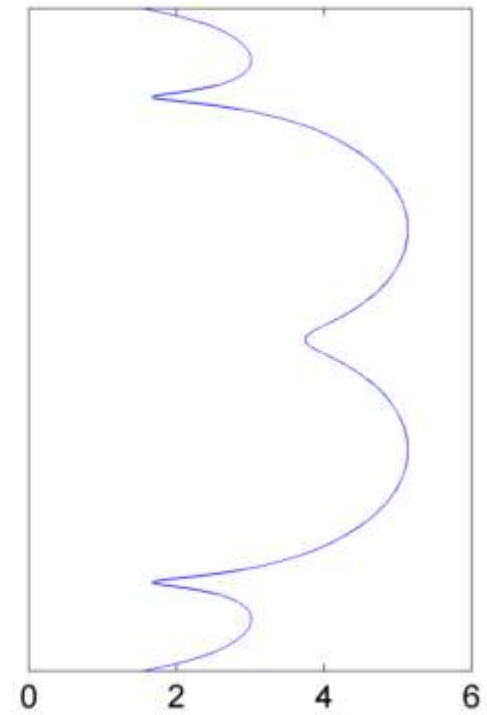
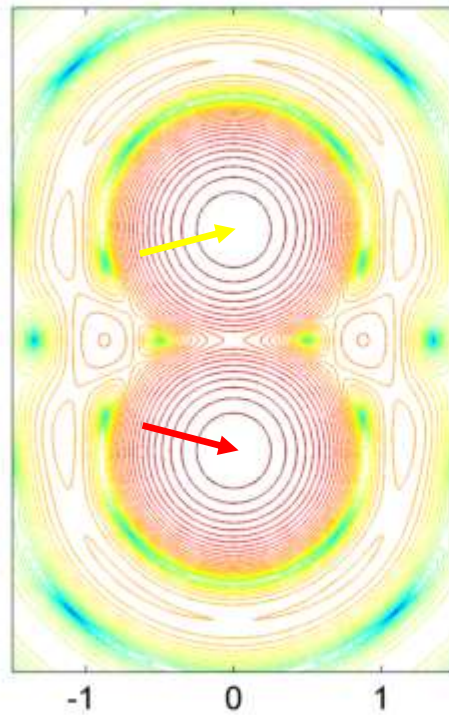
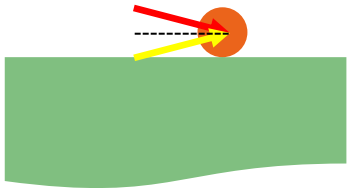
1. Reflection causes an additional incident beam

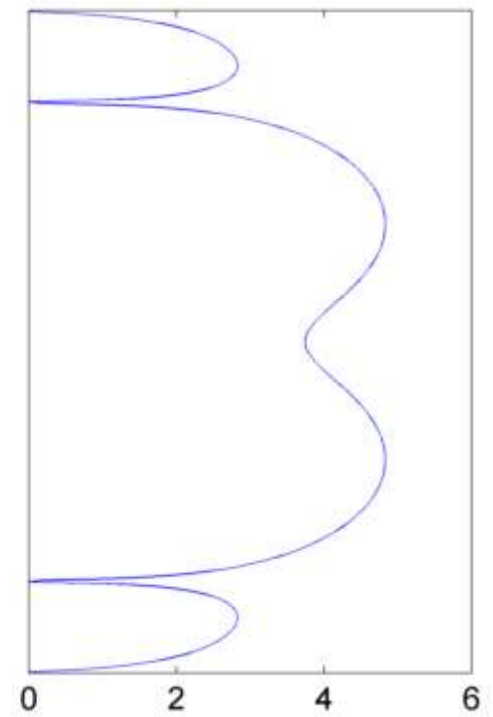
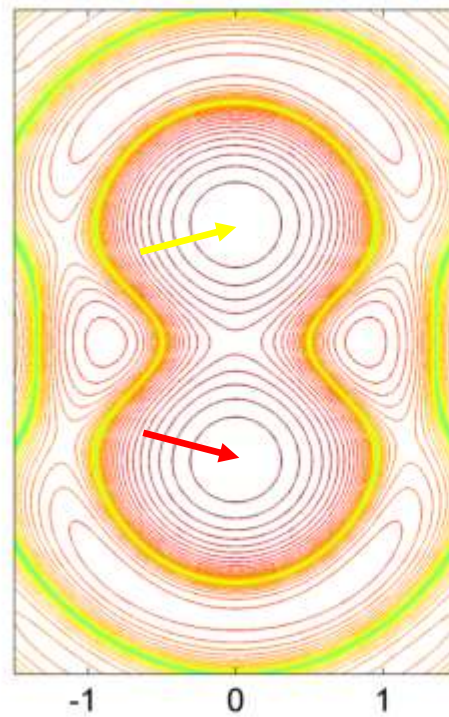


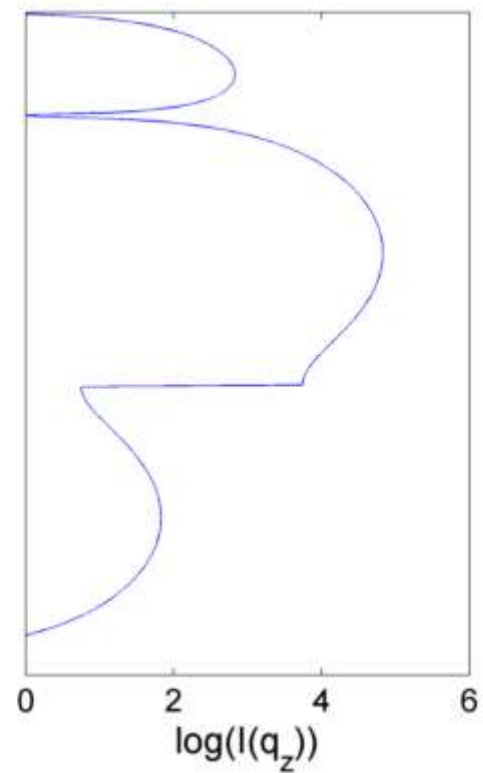
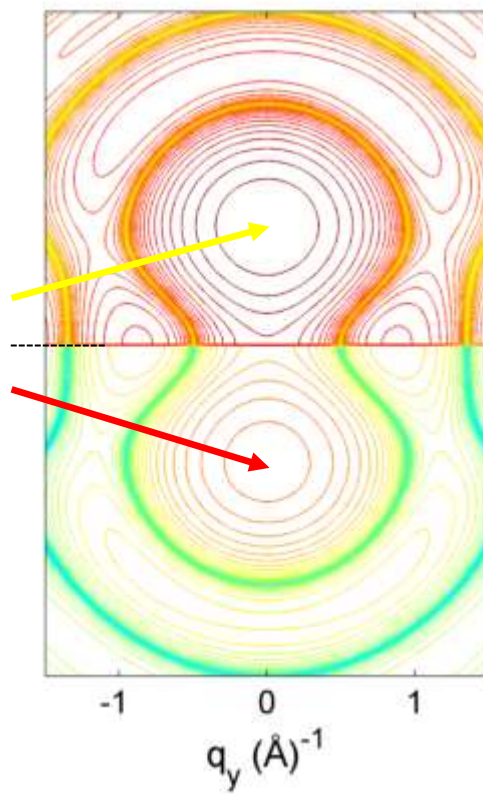
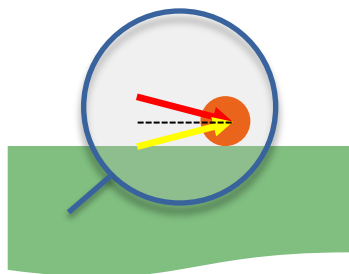




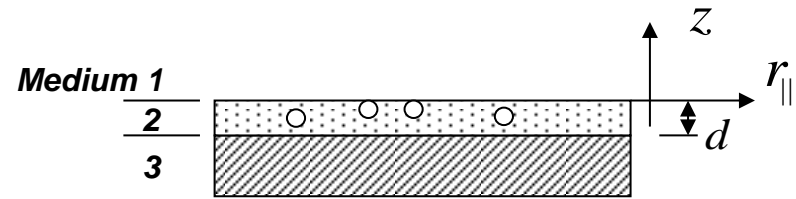






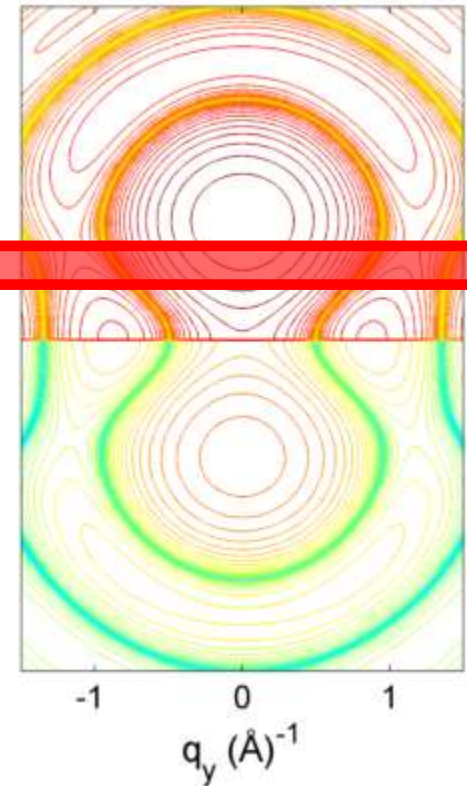
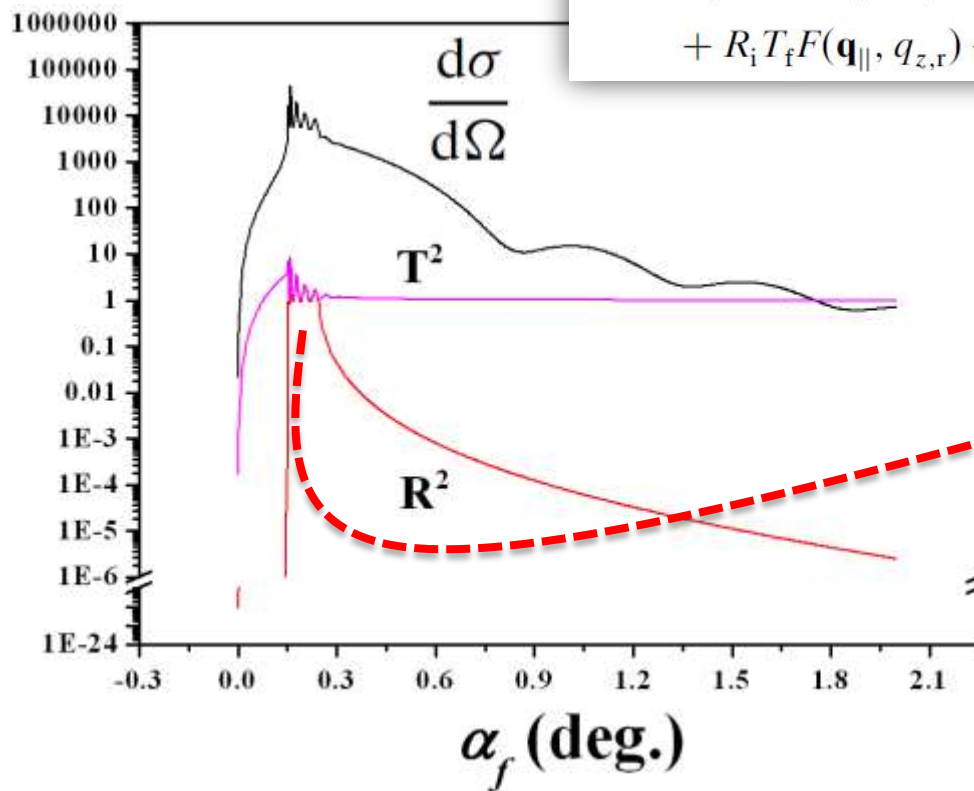


Effect of wave amplitudes

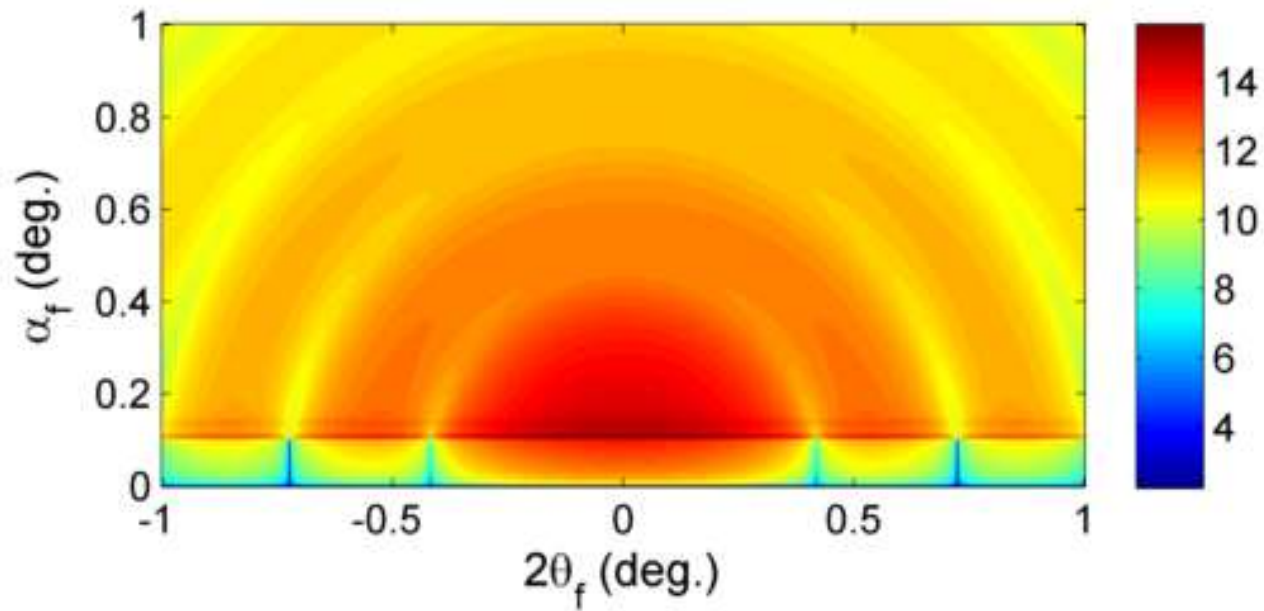


$$\frac{d\sigma}{d\Omega} = r^2 \psi_{sc}(r) \psi_{sc}^*(r)$$

$$= A |T_i T_f F(\mathbf{q}_{\parallel}, q_{z,t}) + T_i R_f F(\mathbf{q}_{\parallel}, -q_{z,r}) + R_i T_f F(\mathbf{q}_{\parallel}, q_{z,r}) + R_i R_f F(\mathbf{q}_{\parallel}, -q_{z,t})|^2$$

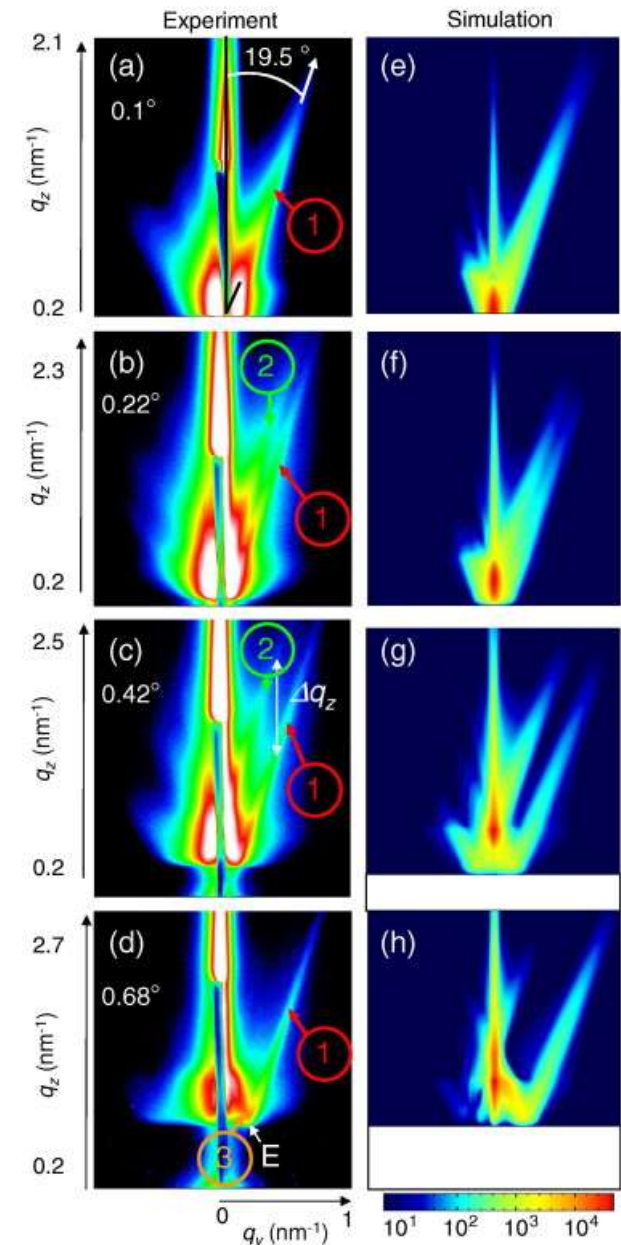
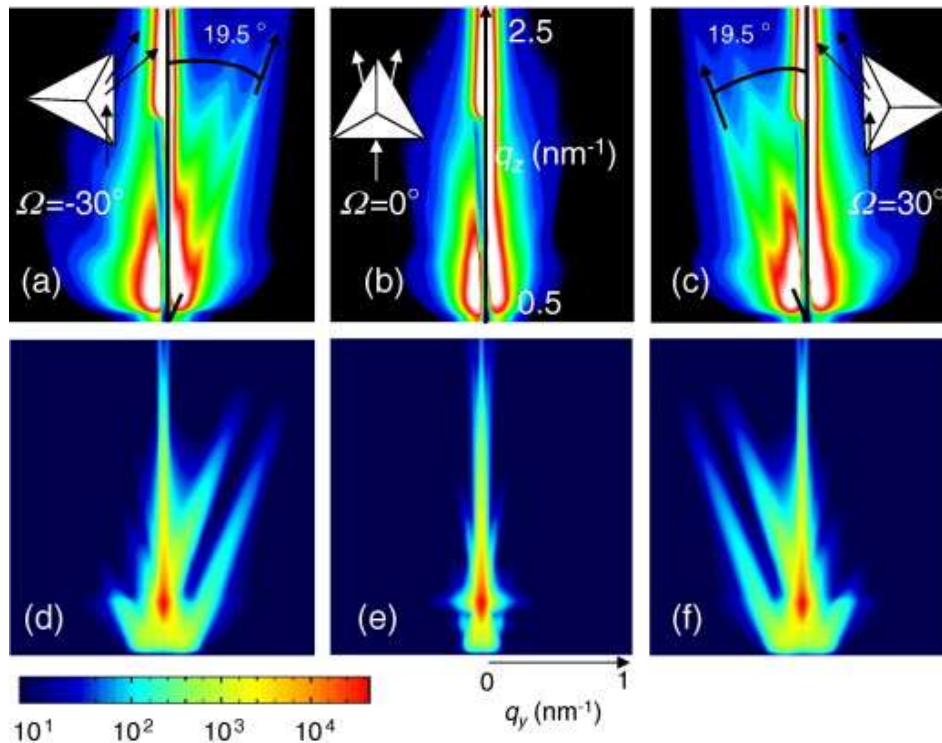


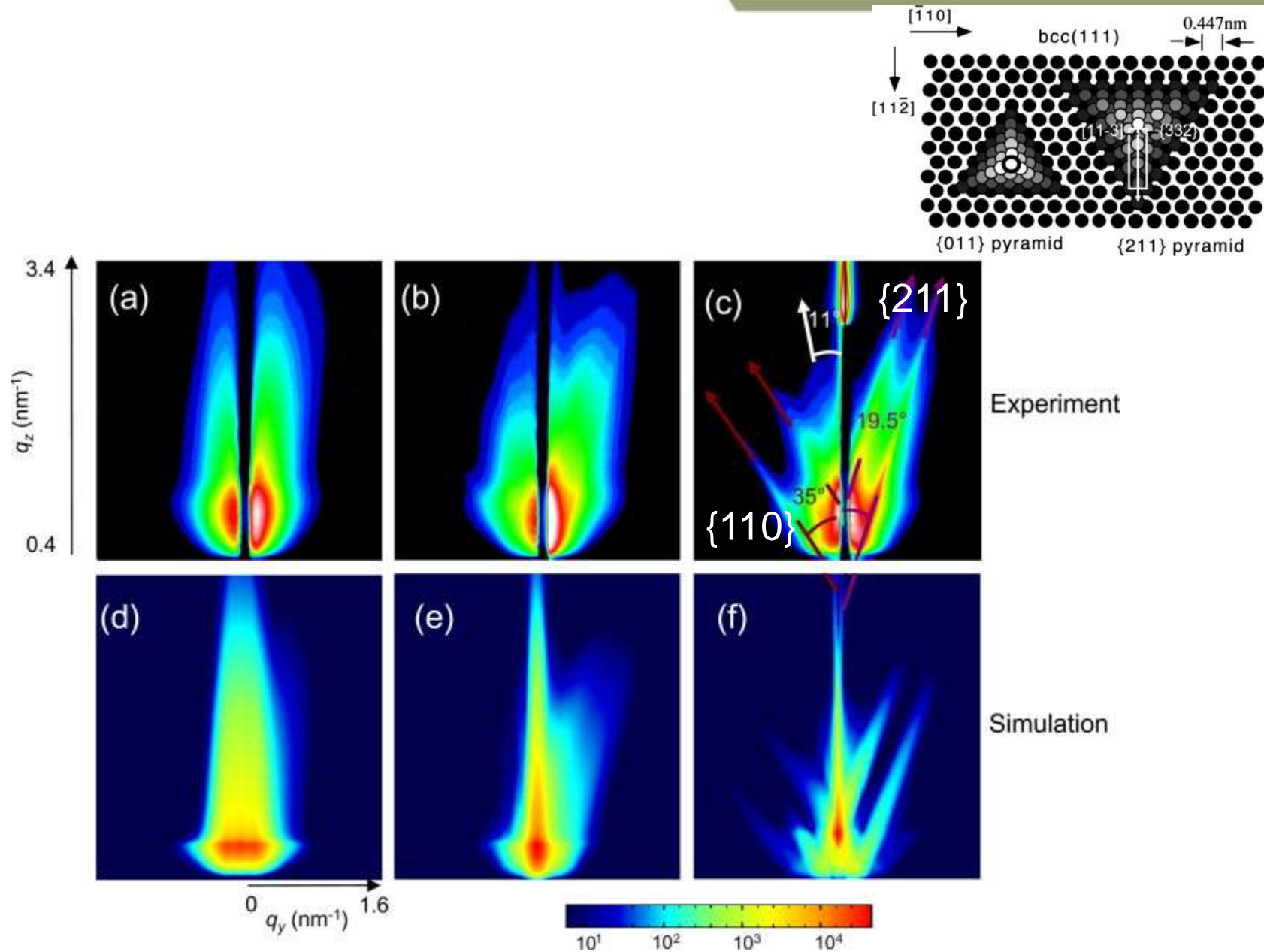
GISAXS from a sphere / log scale image



Example

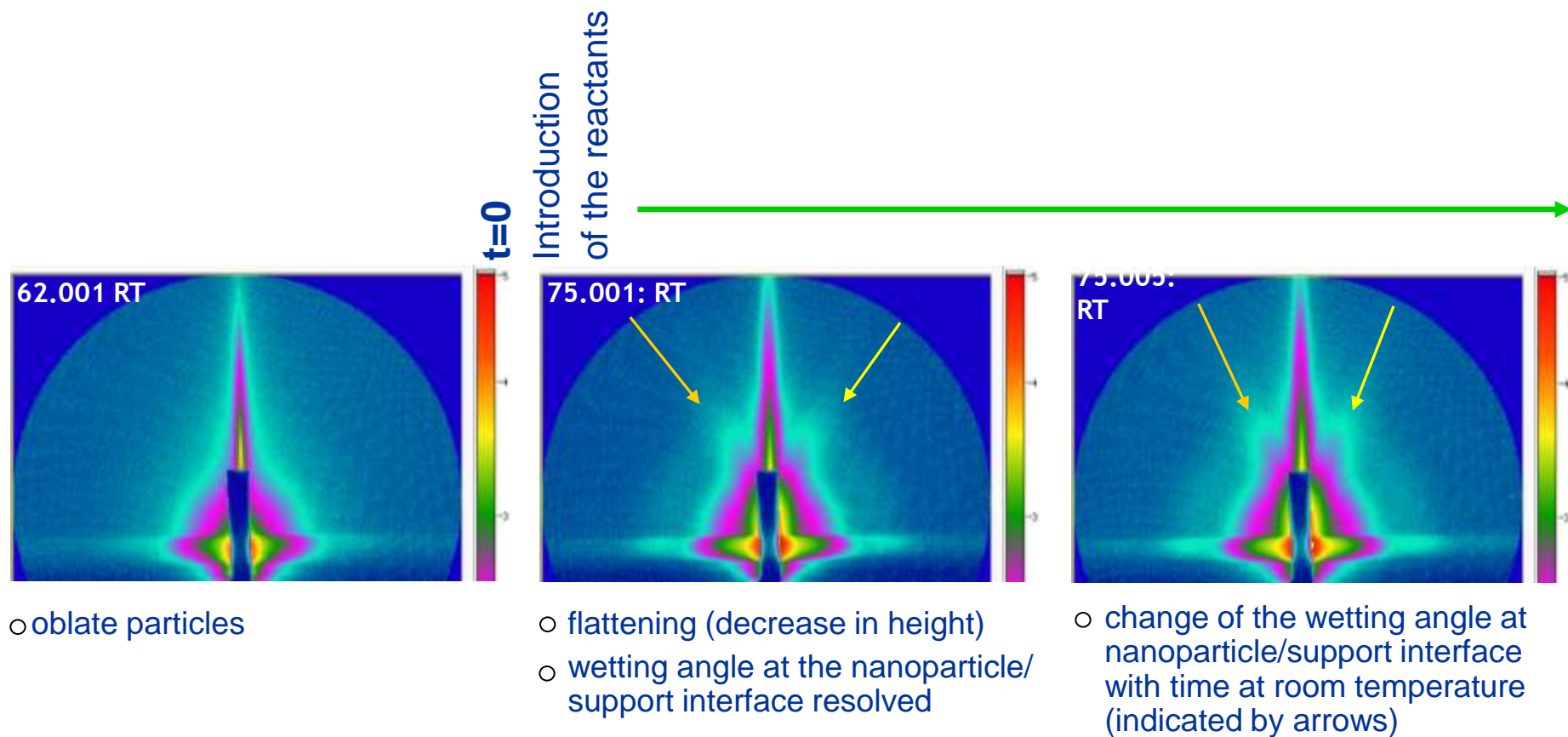
A faceted Pt/W(211) sample annealed at 1340 K





0.8 nm Co/1.1 ML Pt/W after an annealing during (a) 3 min at 920–1000 K, (b) 3 min at 1020–1100 K, (c) 3 min at 1140–1210 K. Respective simulated 2D GISAXS patterns from (d) to (f).

Shape changes with Ag nanoparticles in propylene epoxidation



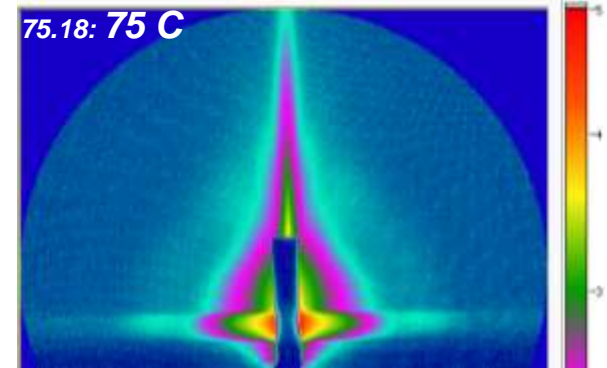
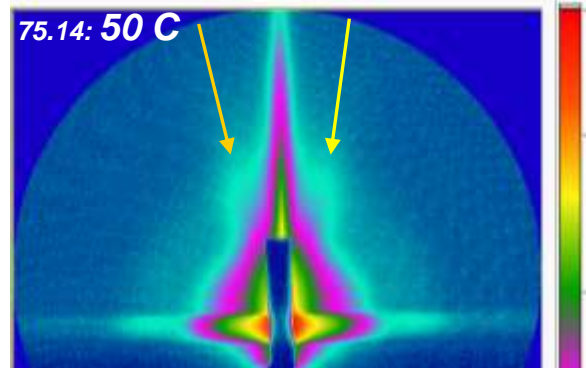
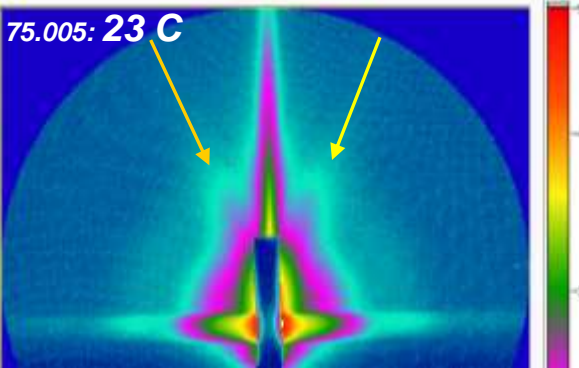
Under ~1 atm. pressure of 1.0% propylene and 0.5% oxygen in He

On 6 cycle Al_2O_3 over SiO_2/Si



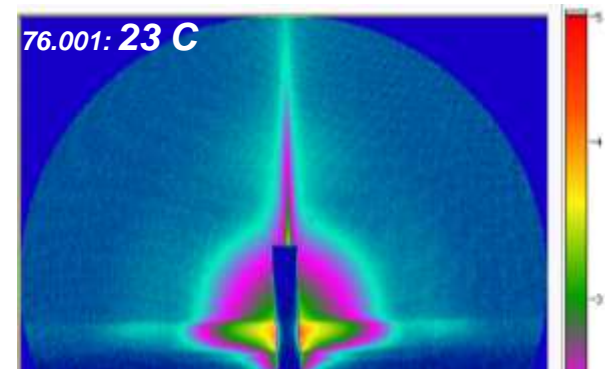
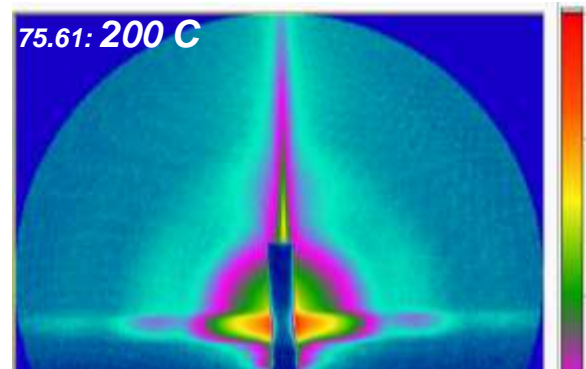
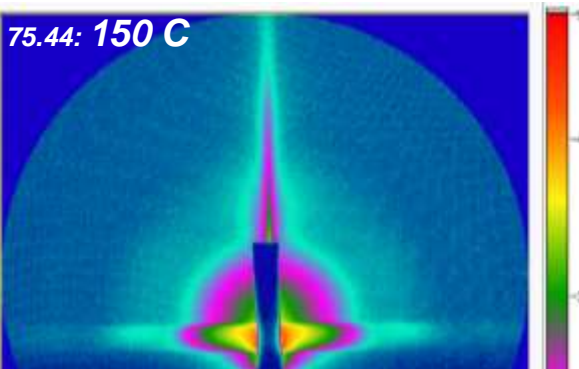
Shape changes with Ag nanoparticles in propylene epoxidation

Under ~1 atm. pressure of 1.0% propylene and 0.5% oxygen in He
On 6 cycle Al₂O₃ over SiO₂/Si with an initial cluster size ~6nm



○ change of the wetting angle

○ onset of change of aspect ratio



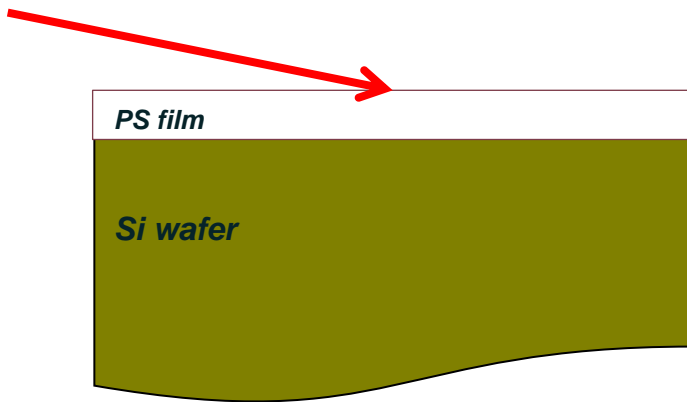
○ particle shape transformation to spherical form

○ particles remain spherical after 4 hrs reaction and cooling back to 23 C

The effect of small incidence angle

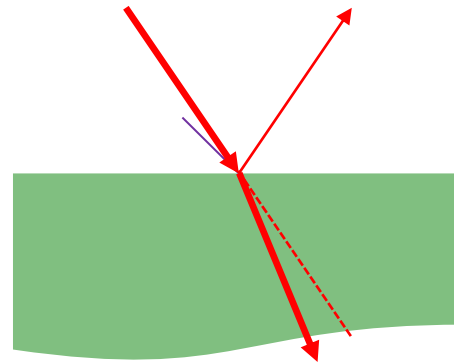
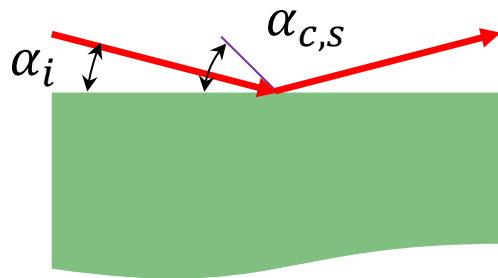
2. Critical angle

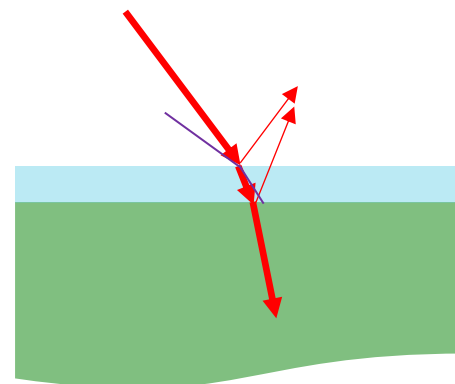
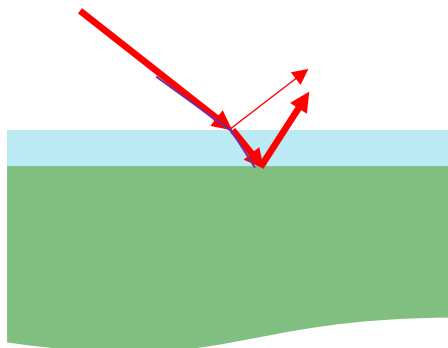
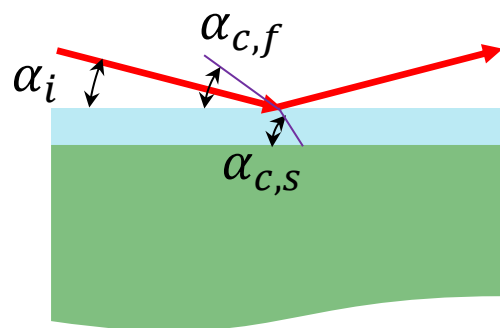
$$\alpha_c = \lambda(r_e \bar{\rho} / \pi)^{1/2}$$

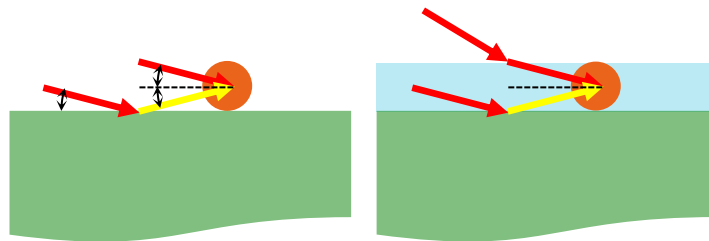


Electron density	Critical angle at 8keV
0.32 e/Å ³	0.150 degree
0.70 e/Å ³	0.223 degree

Critical angles



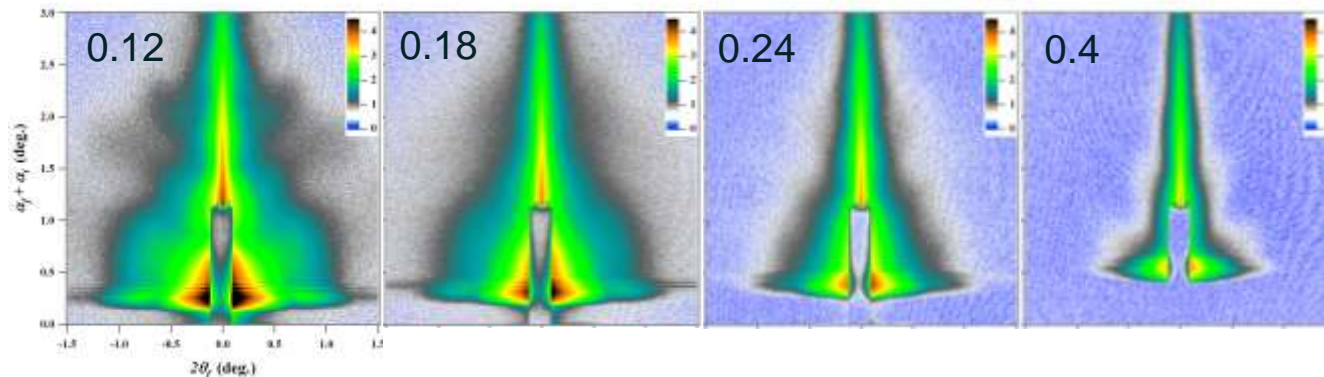
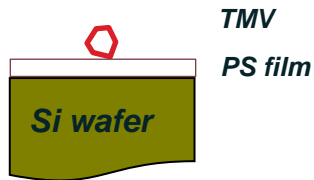




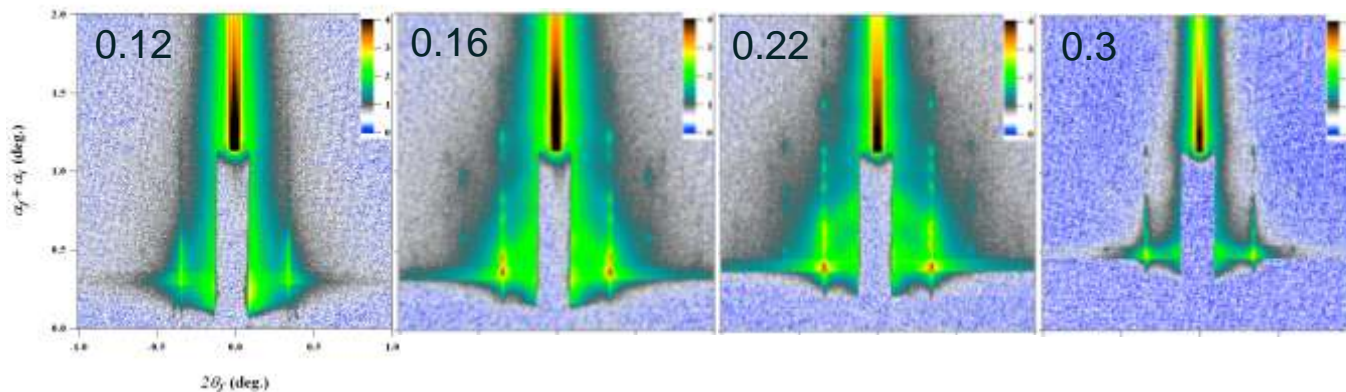
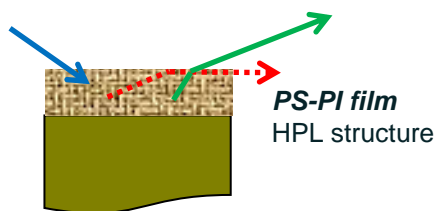
Vary the incident angle!!

X-ray energy: 7.38keV

Critical angles of PS and PS-PI block copolymer film and Si wafer $\sim 0.16^\circ$ and 0.25° , respectively.



As long as the incident angle is smaller than the critical angle of substrate, particle scattering will be detected.
If overshooting is not an issue, the smaller incident angle is the better because smaller q is accessible.



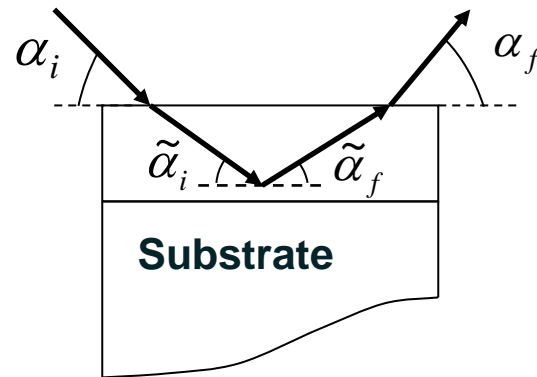
If the incident angle is smaller than the critical angle of film, x-ray can only scan top surface of film
If the exit angle is smaller than the critical angle of film, scattered x-ray practically cannot be detected.

B. Lee et al. **J. Appl. Cryst.** 2008, 41, 134-142.

B. Lee et al. **J. Appl. Cryst.** 2007, 40, 496-504.

The effect of a small incident angle

3. Refraction



$$n_f = 1 - \delta + i\beta \equiv 1 - \alpha_c^2/2 + i\beta$$

$$\alpha_c = \lambda(r_e \bar{\rho}/\pi)^{1/2}$$

$$n_f \cos \tilde{\alpha} = \cos \alpha$$

How much x-ray will be refracted in a polymer film

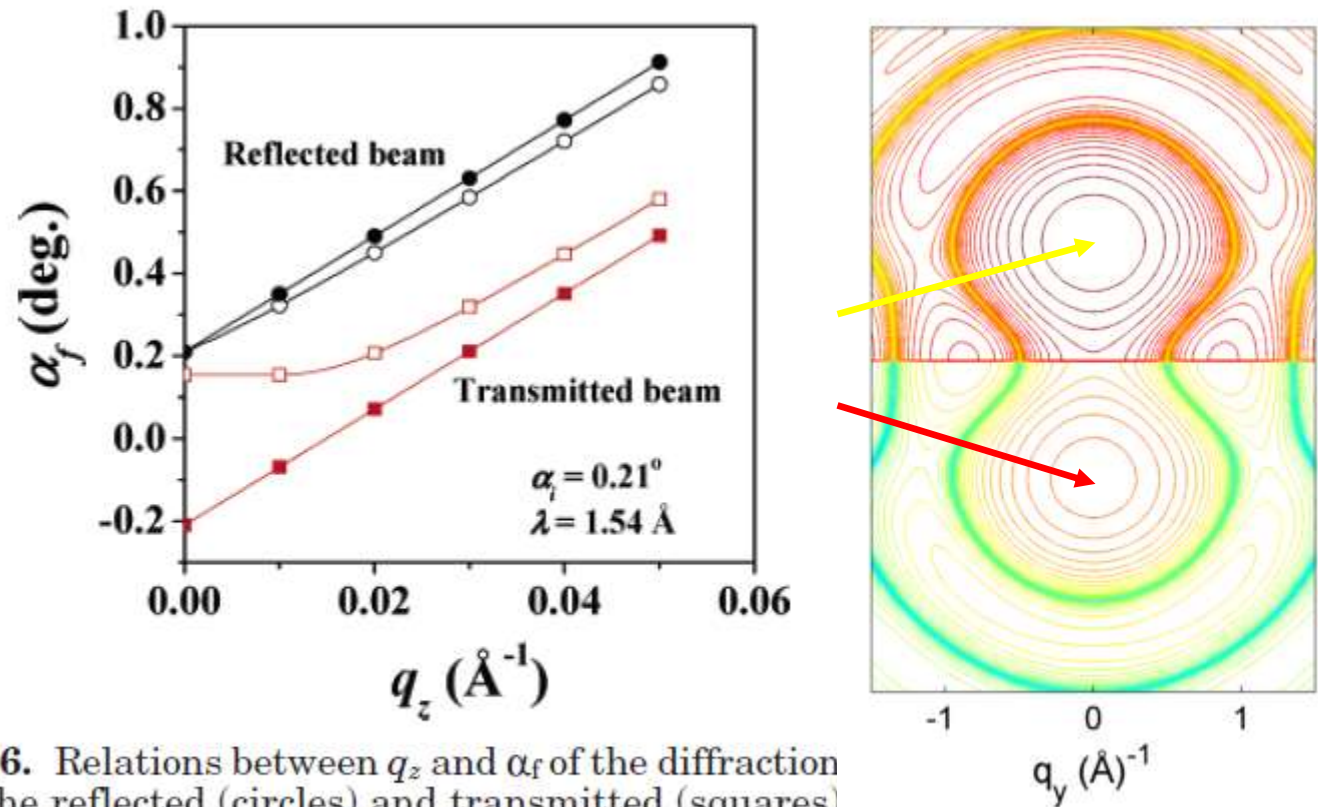
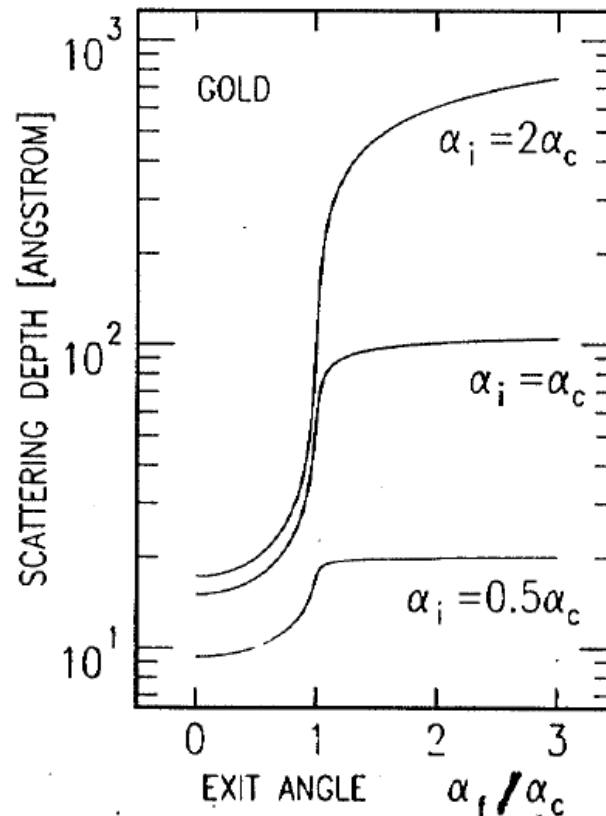


Figure 6. Relations between q_z and α_f of the diffraction due to the reflected (circles) and transmitted (squares) beams: the filled and open symbols denote the diffraction peaks without and with correction for the refraction effect, respectively.

B. Lee et al. Macromolecules, 2005, 38 (10), pp 4311–4323

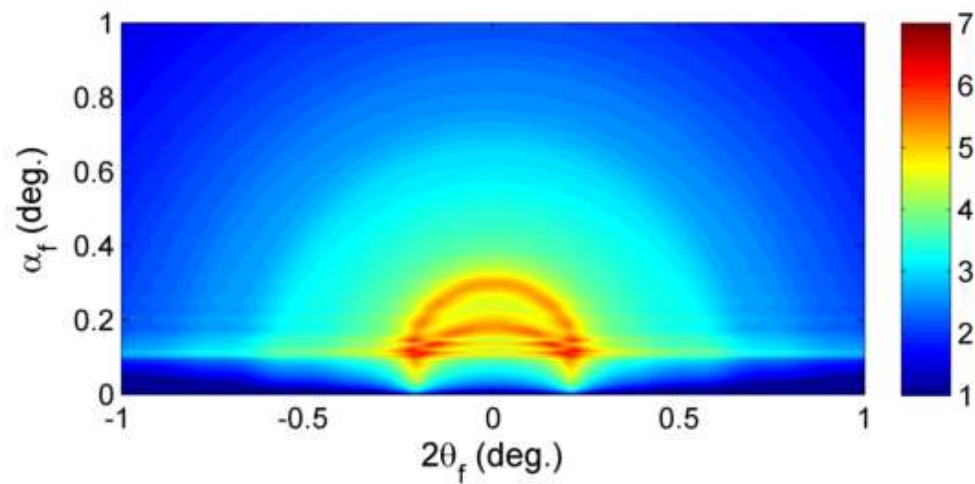
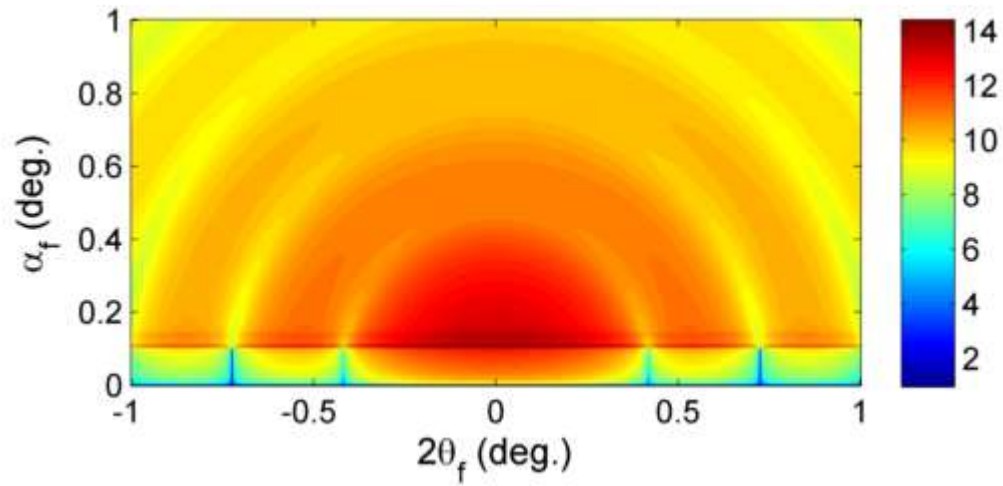
Penetration depth

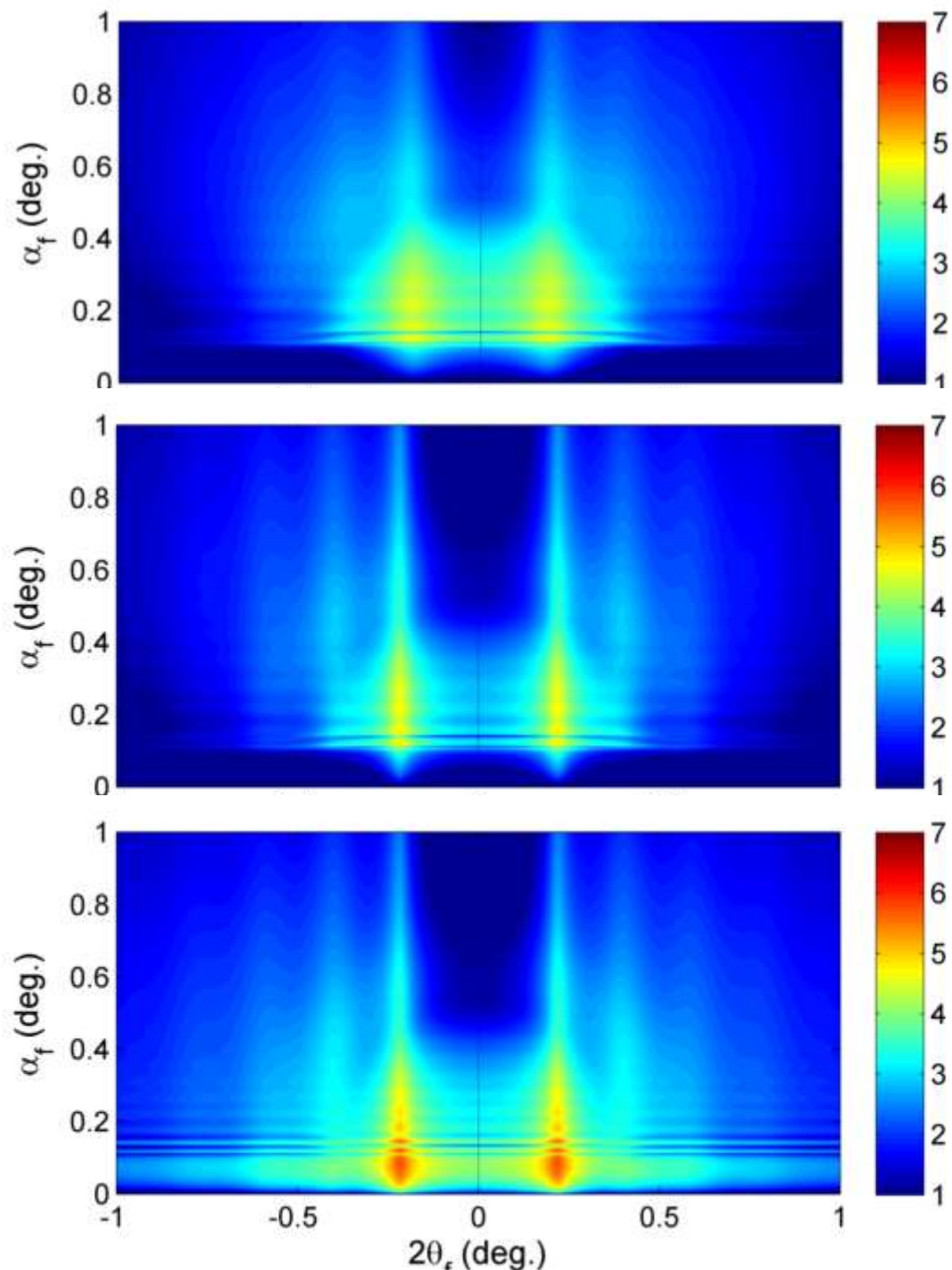


$$\alpha_c = \lambda(r_e \bar{\rho} / \pi)^{1/2}$$

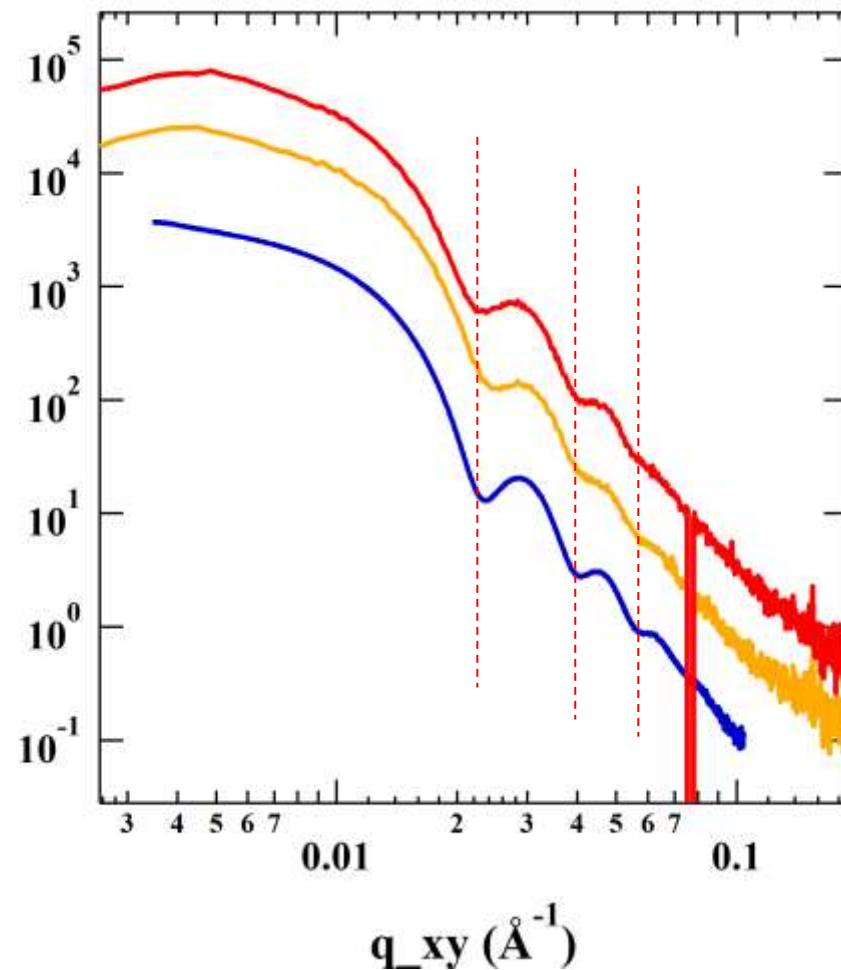
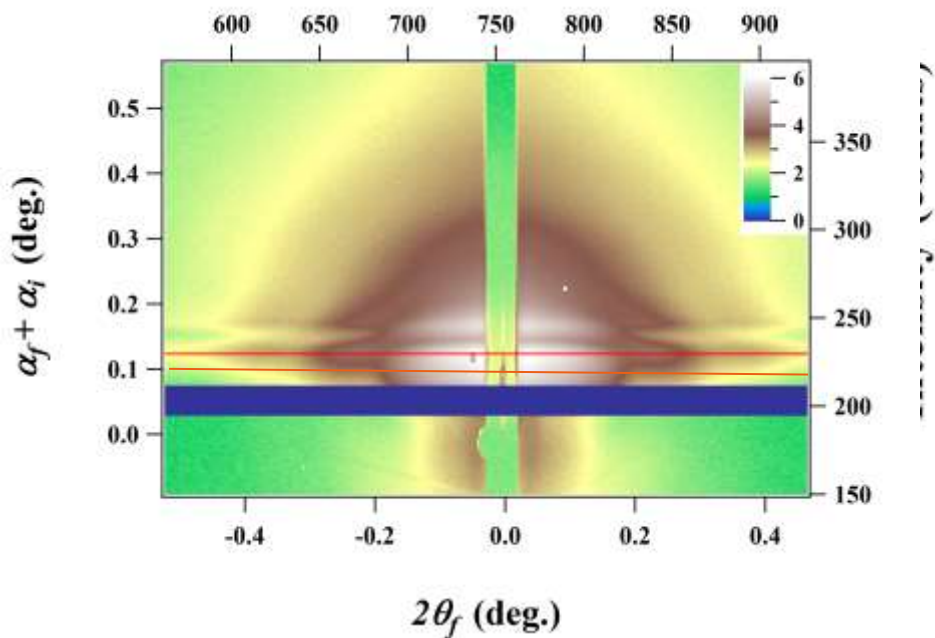
Figure 21. Universal curves of the variation of scattering depth Λ with incidence and exit angles α_i and α_f , expressed as multiples of α_c . The units of the vertical scale depend in general on the photoelectric absorption of the sample, but have been calculated here for gold at $\lambda = 1.5 \text{ \AA}$ using equation (48).

Quiz



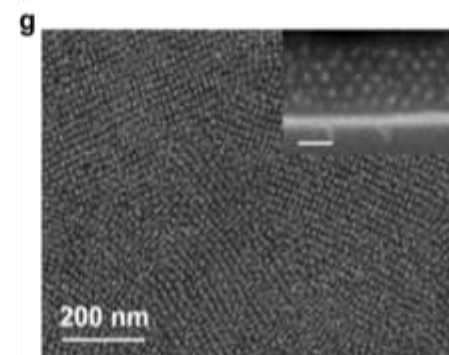
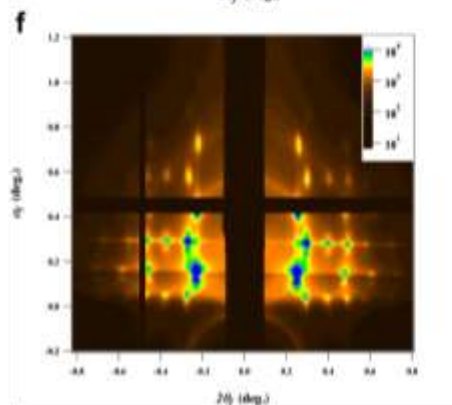
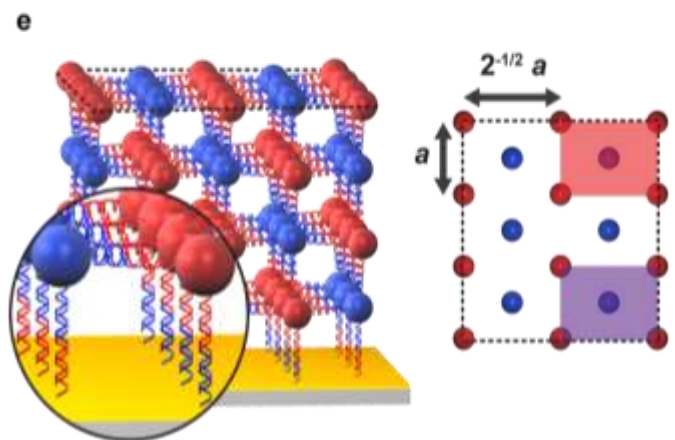
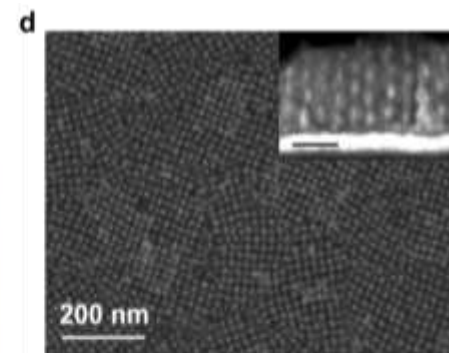
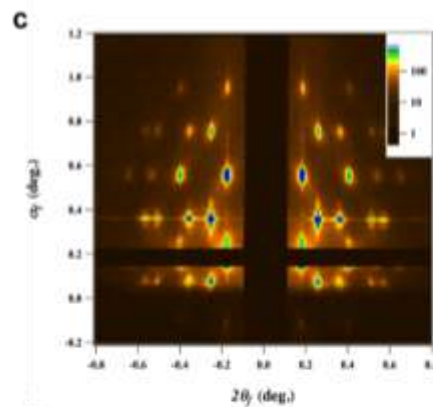
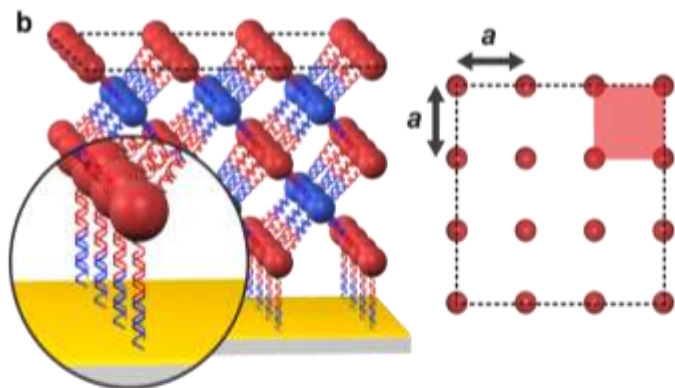


GISAXS vs SAXS : 40nm AuNp

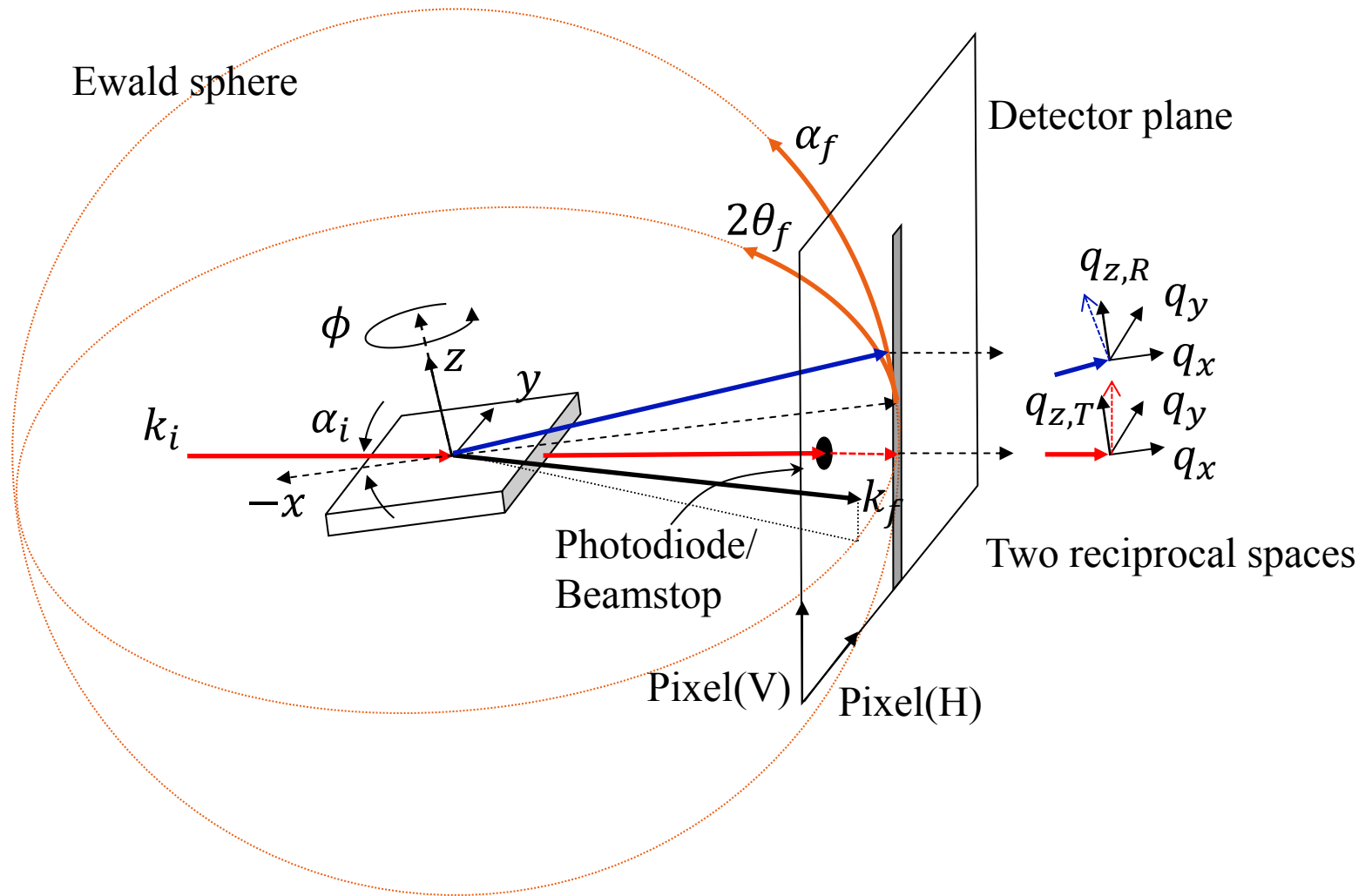


Diffraction from lattice

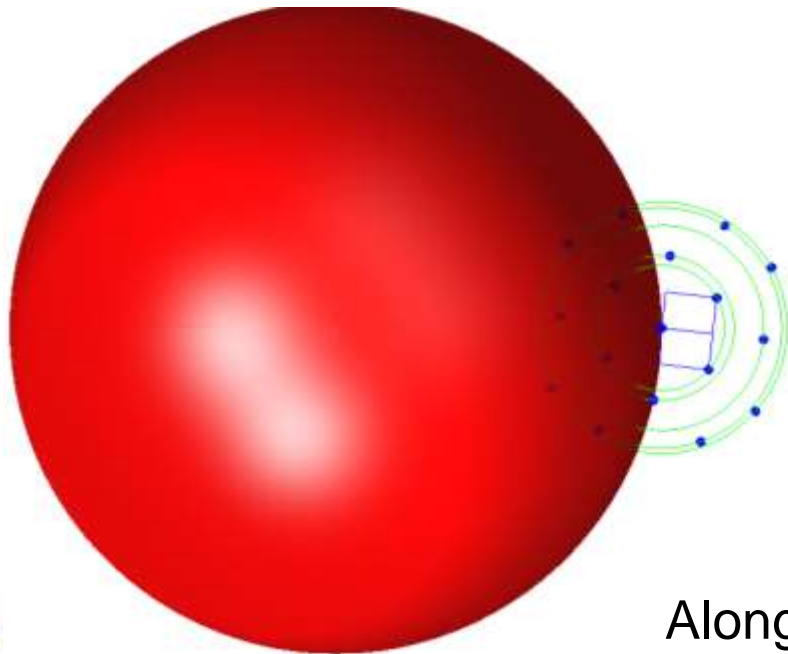
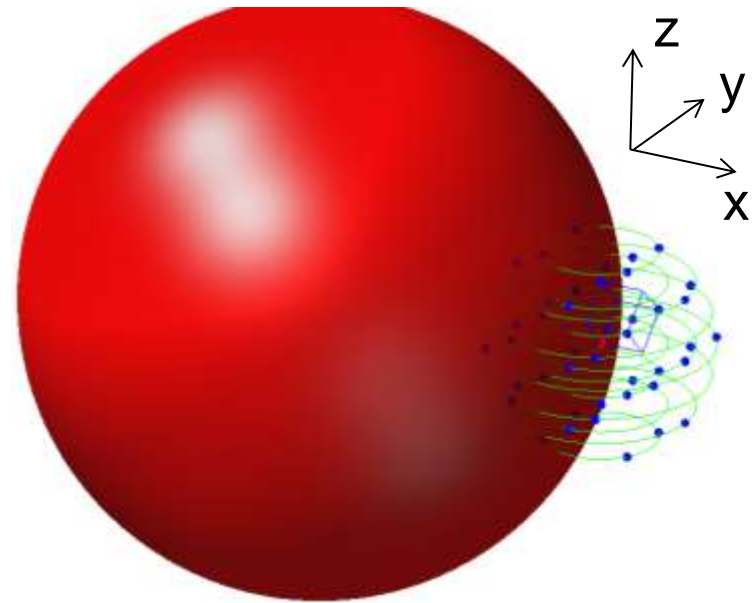
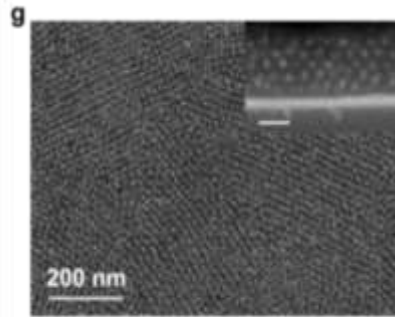
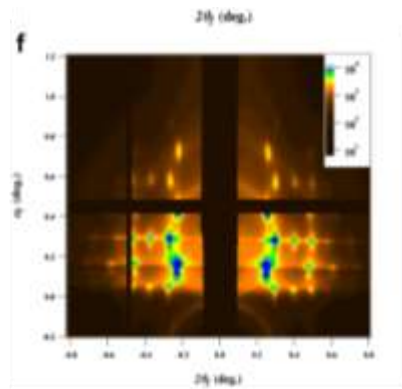
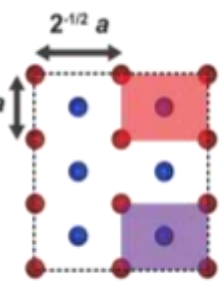
2D powder BCC - 100 vs 110 orientation



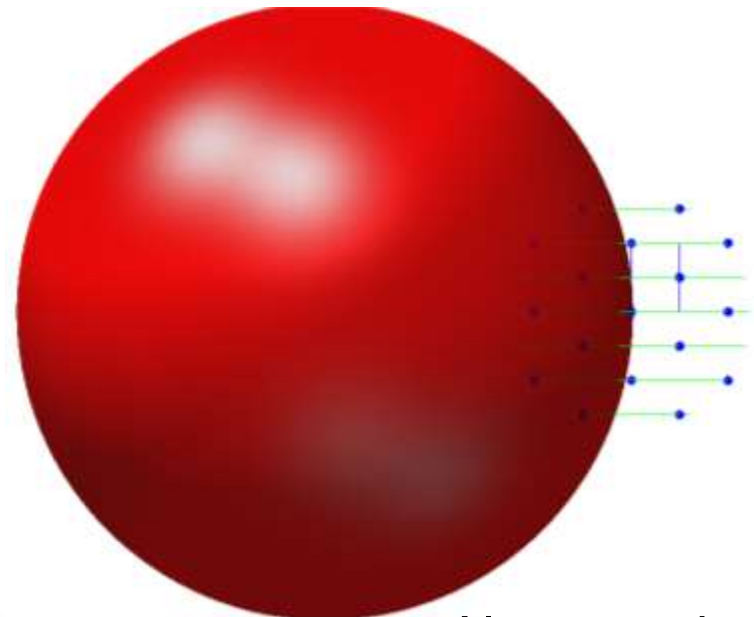
A. Senesi, B. Lee et al. **Angew. Chem. Int. Ed.**, 2013, 52(26), 6624–6628



BCC 110 orientation

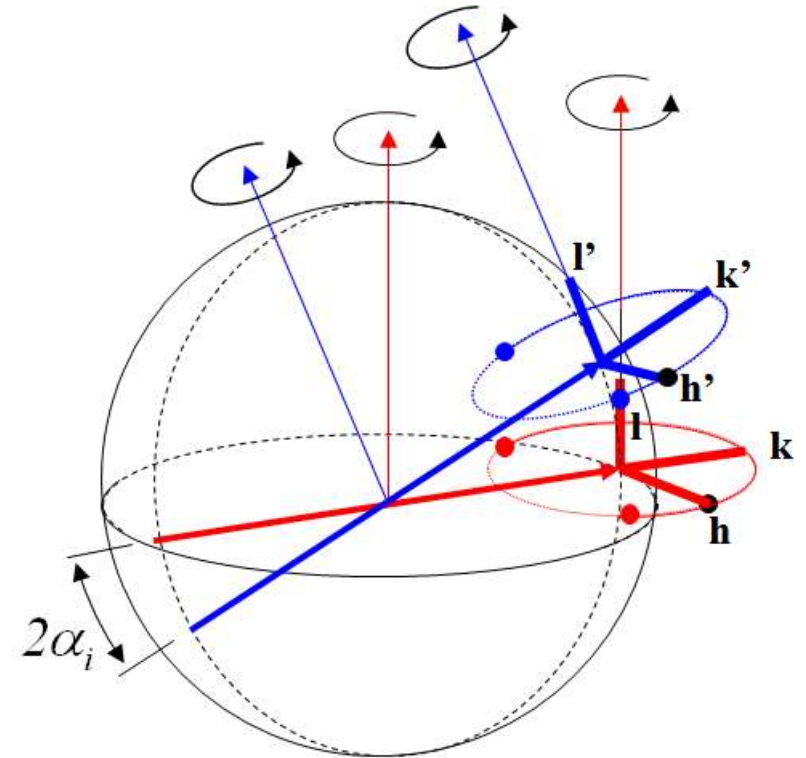
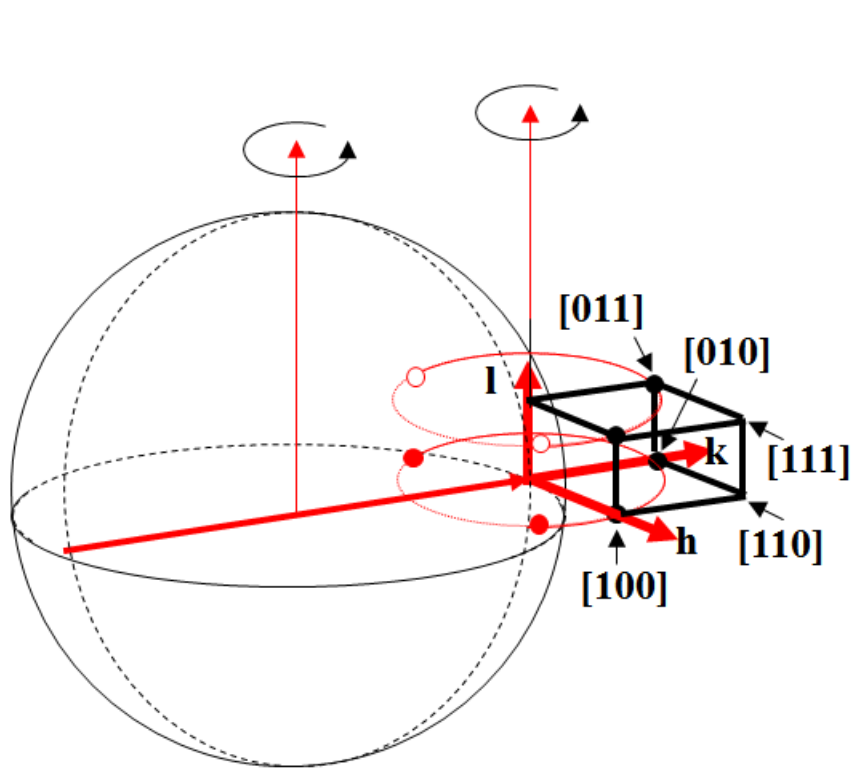


Along z axis



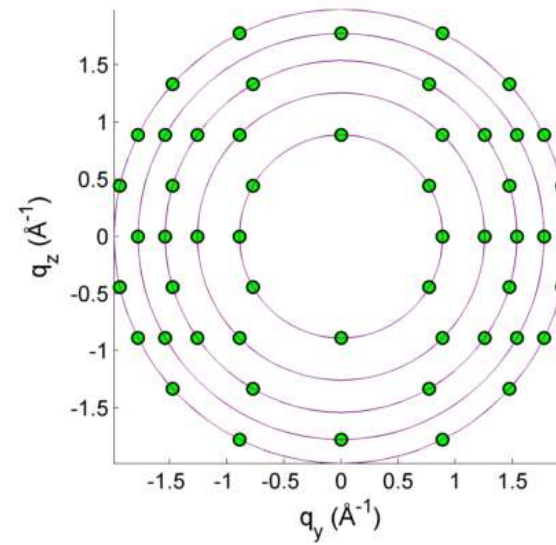
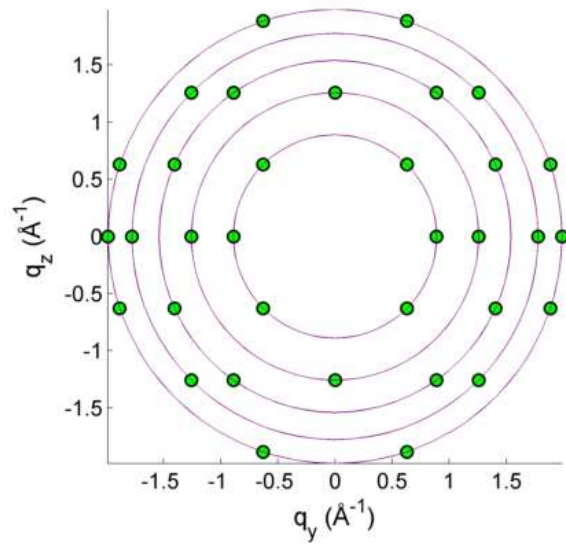
Along y axis

Ewald sphere

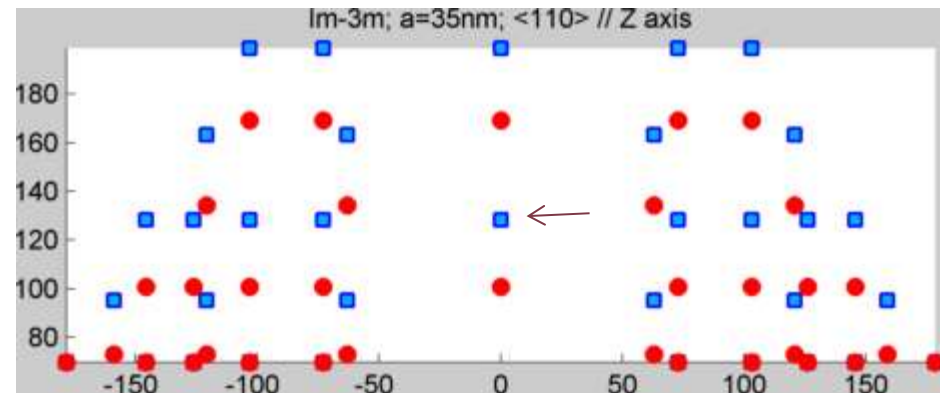
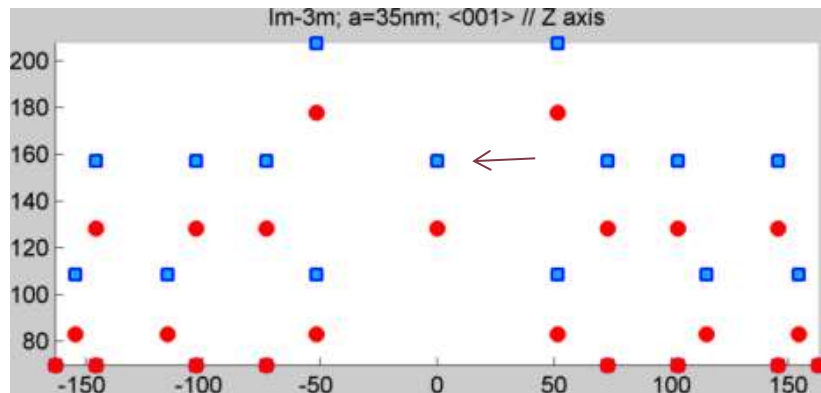


Two-beam effect
Refraction correction
Crystal orientation matrix
Penetration depth

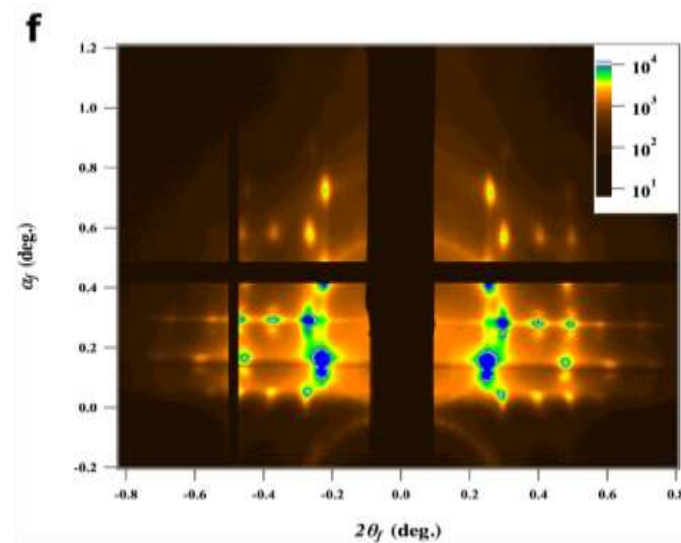
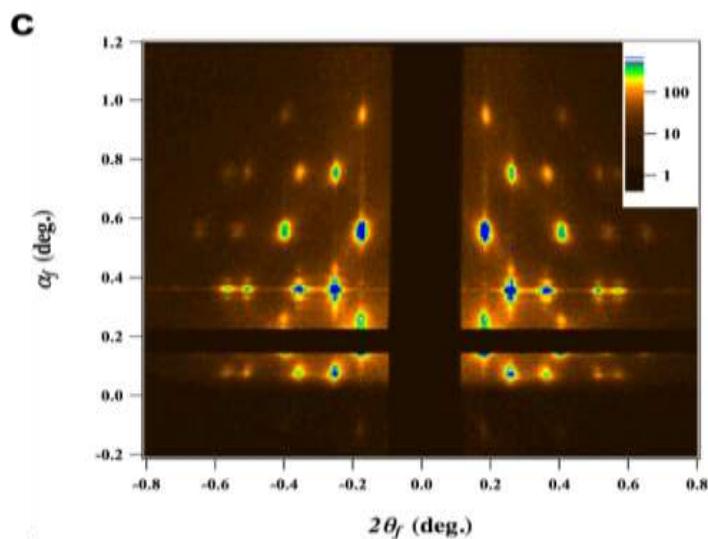
BCC : 100 vs 110



BCC : 100 vs 110

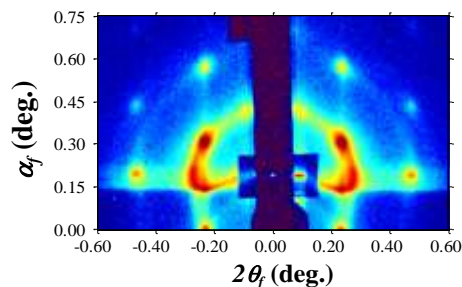


BCC : 100 vs 110

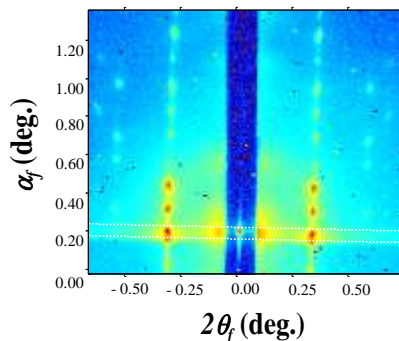


A. Senesi, B. Lee et al. **Angew. Chem. Int. Ed.**, 2013, 52(26), 6624–6628

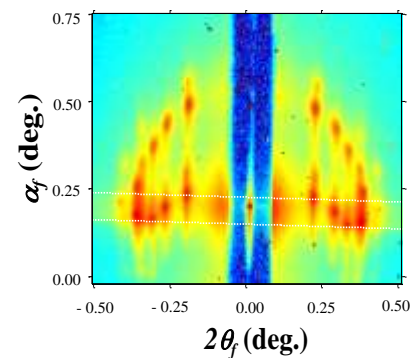
Diffractions from block copolymer films



Hexagonal Cylinder

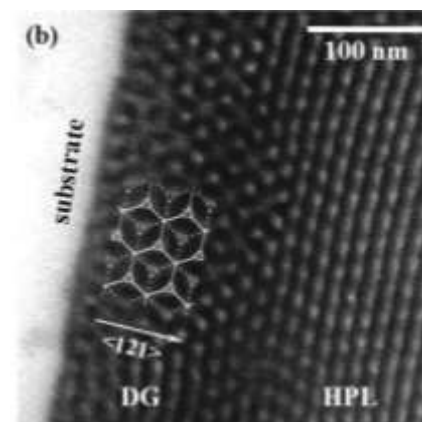
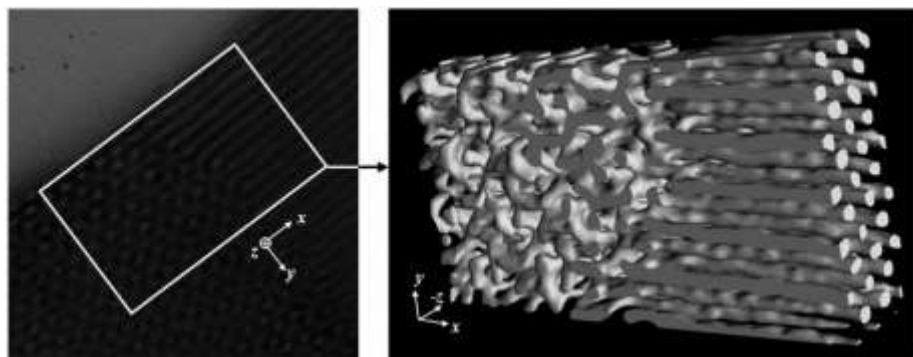


Hexagonally perforated layer



Gyroid : Cubic(*Ia3d*)

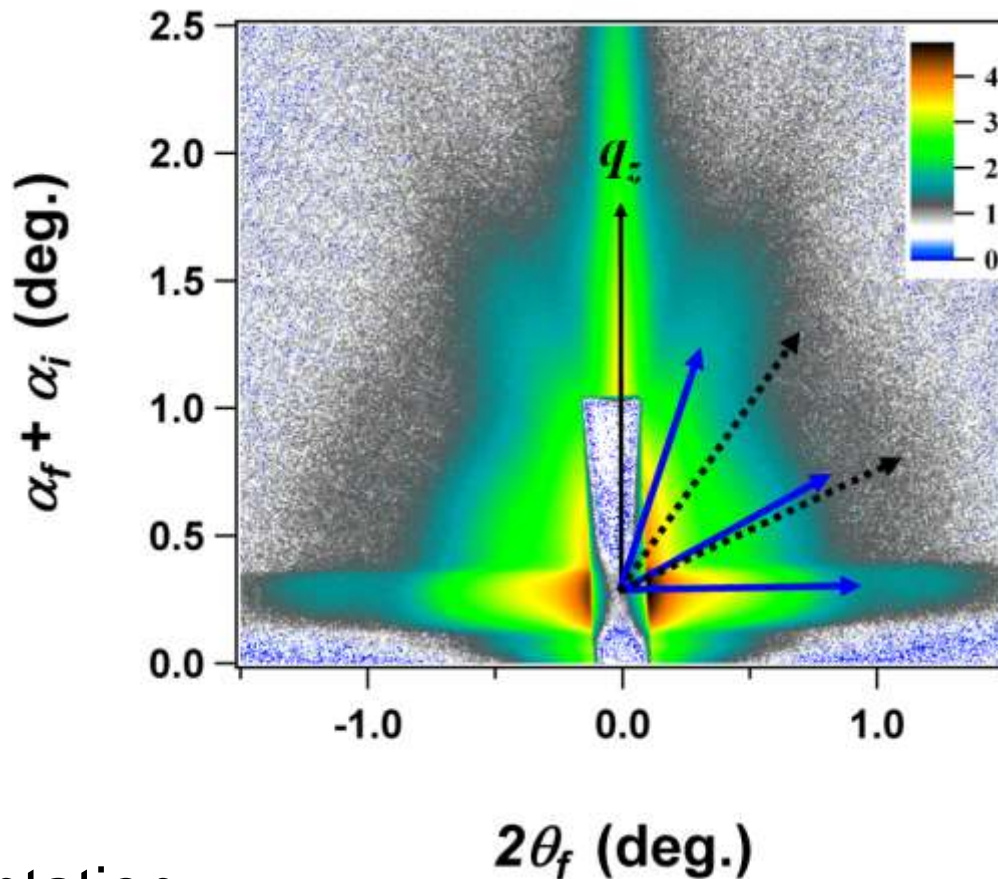
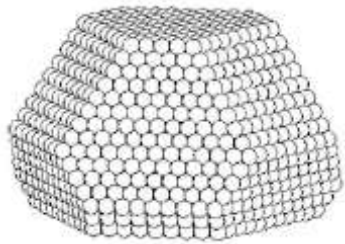
Lee et al. *Macromolecules*, 2005, 38, 4311



H.-W. Park et al. *J. Am. Chem. Soc.*, **2009**, 131 (1), pp 46–47

H.-W. Park et al. *Macromolecules*, 2007, 40 (7), pp 2603–2605

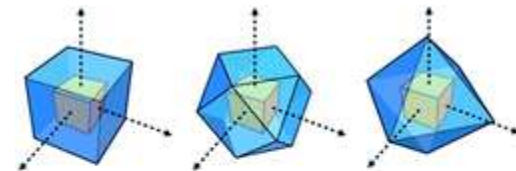
Facet analysis



112 orientation

111 : 19.5, 61.9, and 90°

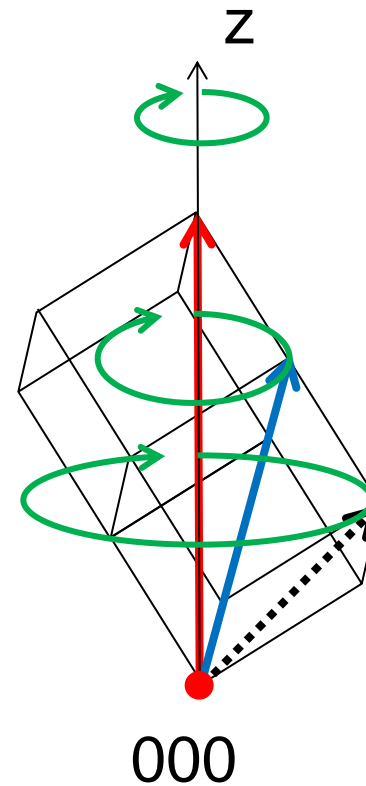
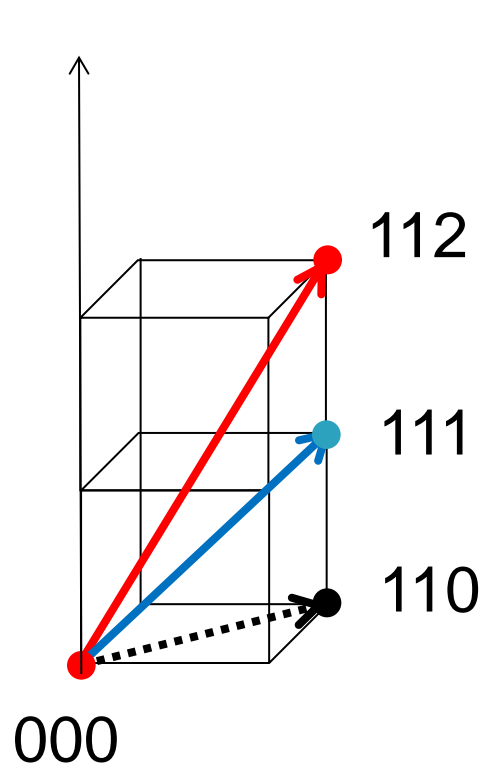
110 : dotted arrows



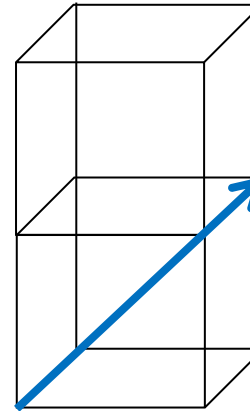
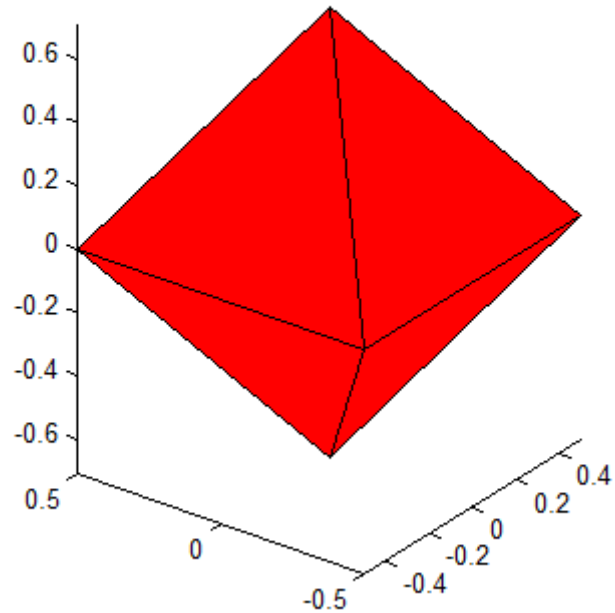
Catalytic Nanoparticles Shaping Up!



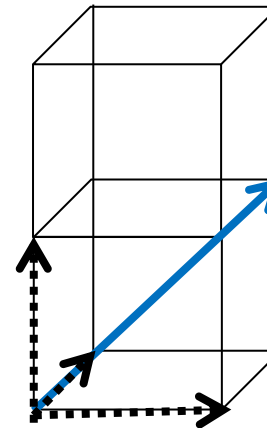
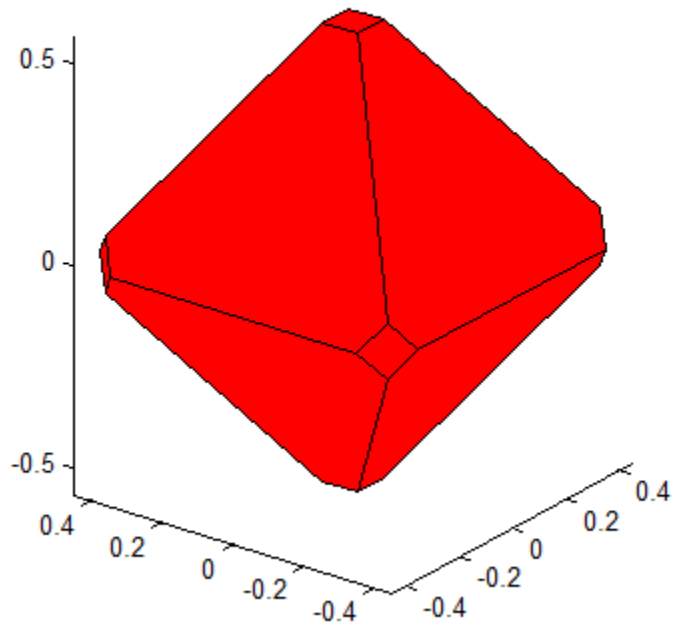
2D powder with 112 orientation



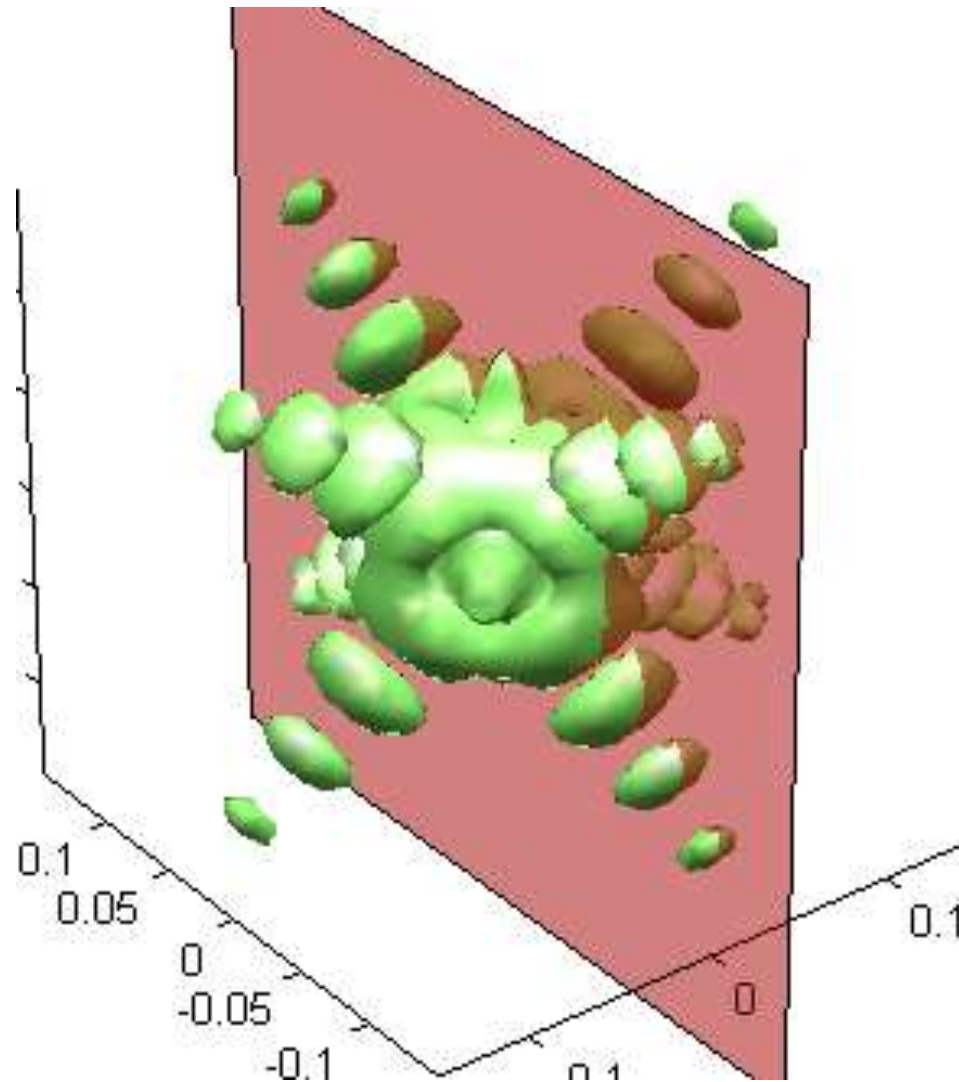
Octahedron



Truncated Octahedron

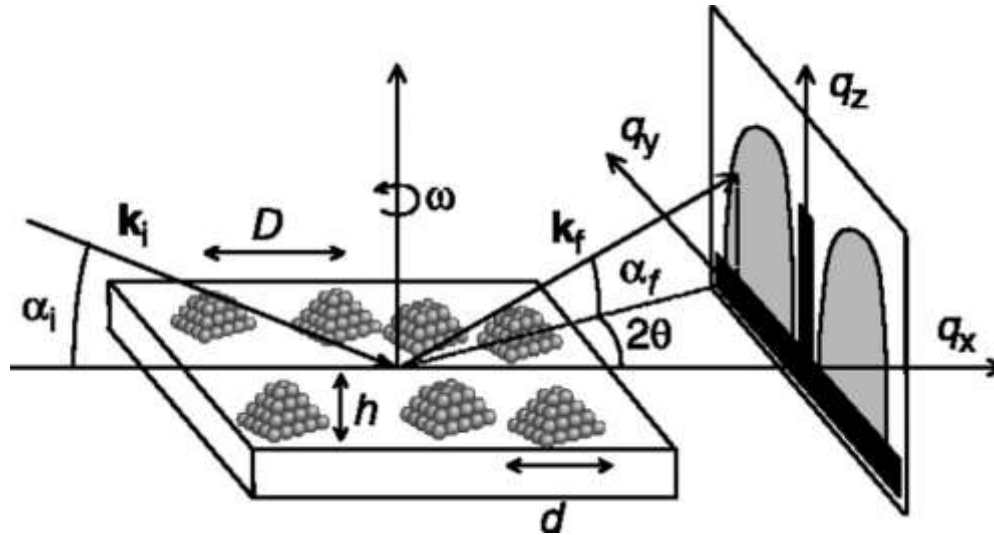


Octahedron scattering in the reciprocal space



Quantitative calculation



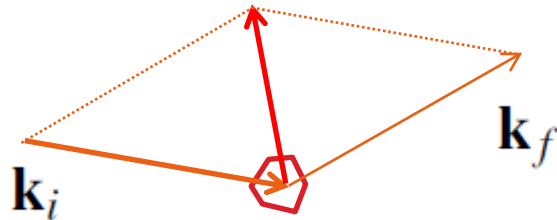


$$\begin{pmatrix} q_x \\ q_y \\ q_{z,X} \end{pmatrix} = n_f k_0 \begin{pmatrix} \cos \tilde{\alpha}_f \cos 2\theta_f - \cos \tilde{\alpha}_i \\ \cos \tilde{\alpha}_f \sin 2\theta_f \\ \sin \tilde{\alpha}_f \pm \sin \tilde{\alpha}_i \end{pmatrix}$$

where $X = t$ or r for $+$ or $-$, and the tilde indicates the refracted angle.

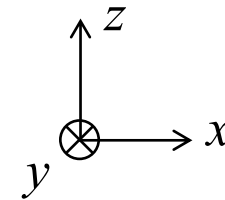
Definition of q : Four q 's in GISAXS due to the reflection

$$\mathbf{q} = \mathbf{k}_f - \mathbf{k}_i$$



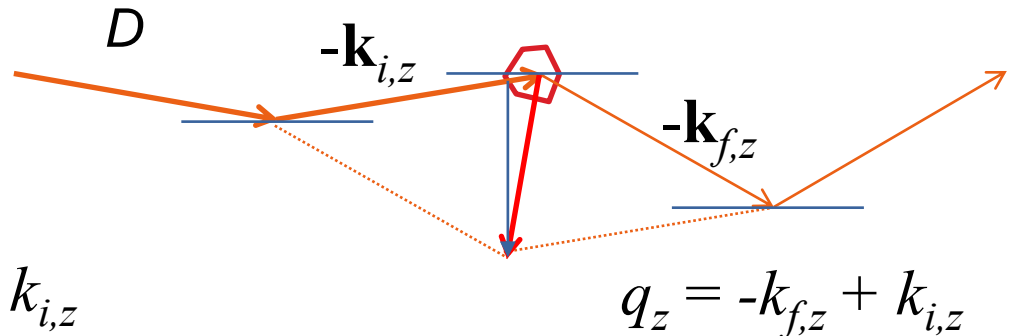
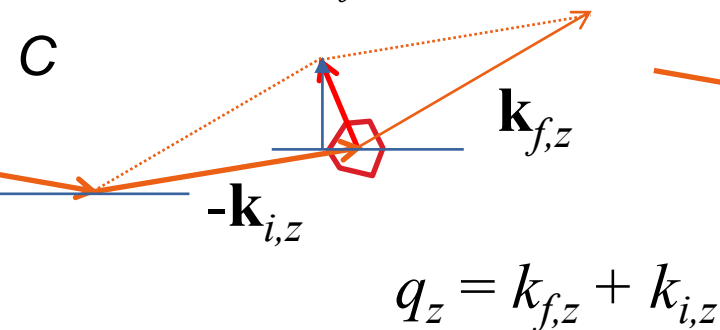
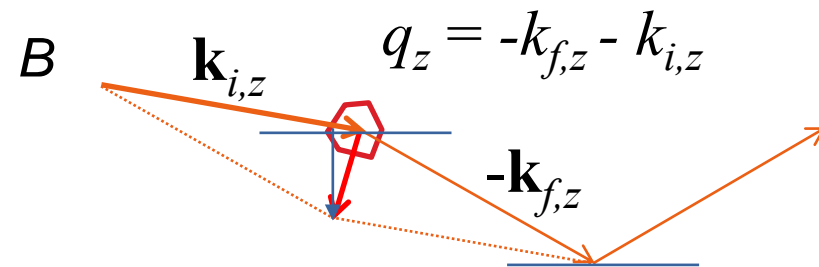
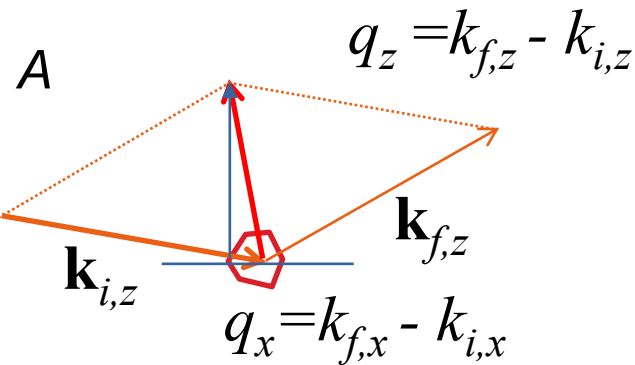
$$q_z = k_{f,z} - k_{i,z}$$

$$q_x = k_{f,x} - k_{i,x}$$



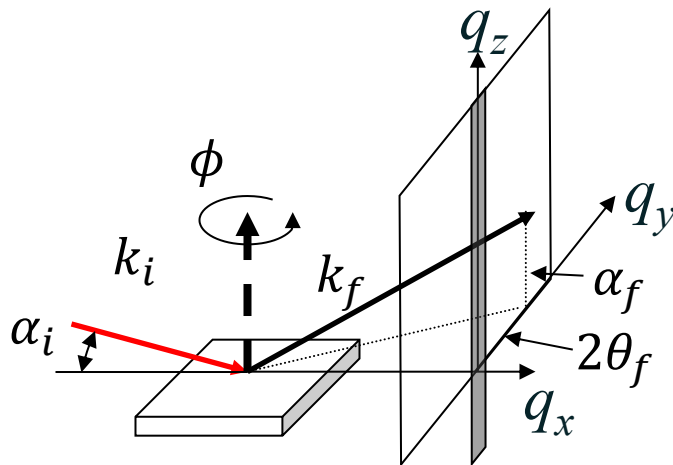
SAXS

GISAXS



Full DWBA formulae

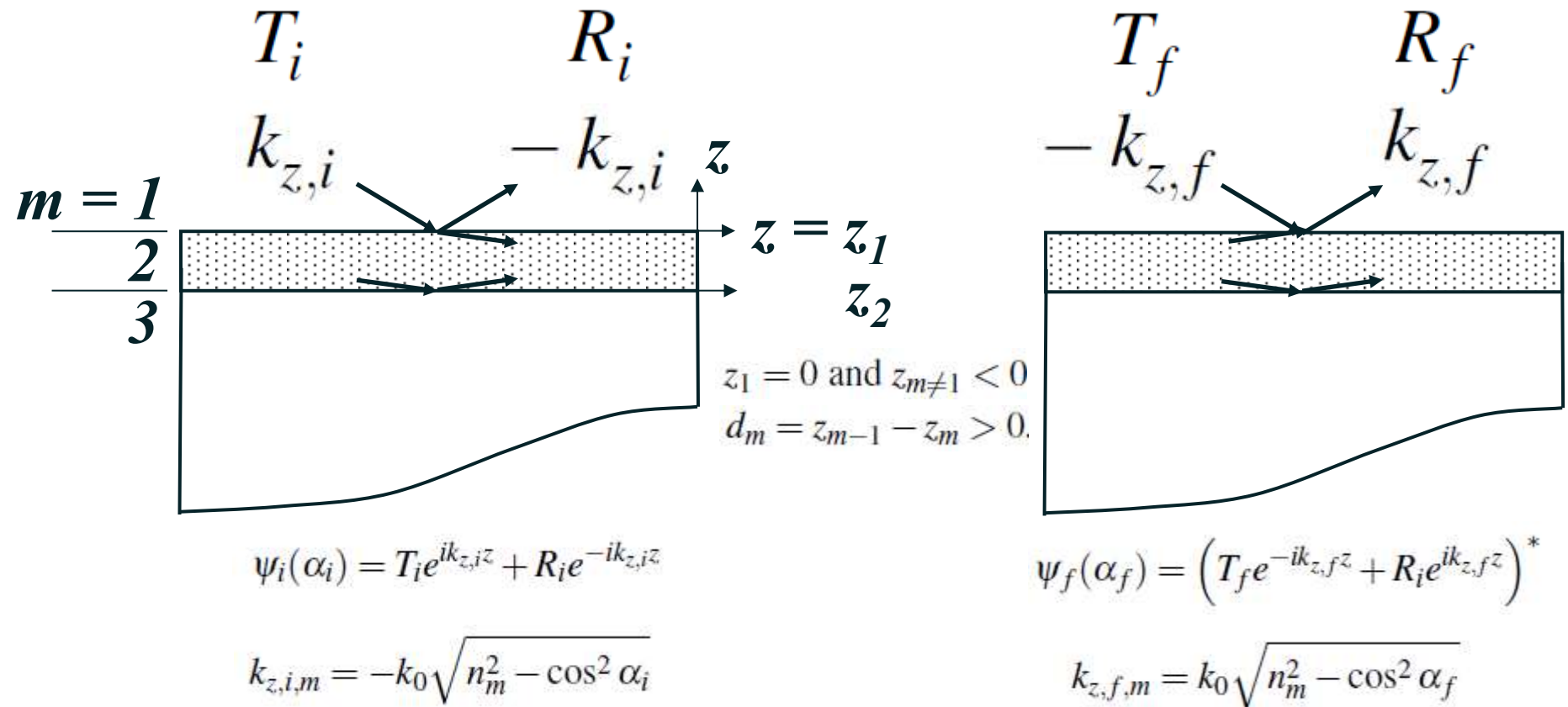
$$\begin{aligned}\frac{d\sigma}{d\Omega} &= r^2 \psi_{\text{sc}}(r) \psi_{\text{sc}}^*(r) \\ &= A \left| T_i T_f F(\mathbf{q}_{||}, q_{z,t}) + T_i R_f F(\mathbf{q}_{||}, -q_{z,r}) \right. \\ &\quad \left. + R_i T_f F(\mathbf{q}_{||}, q_{z,r}) + R_i R_f F(\mathbf{q}_{||}, -q_{z,t}) \right|^2\end{aligned}$$



$$\mathbf{q}_t = (\mathbf{q}_{||}, q_{z,t})$$

$$\mathbf{q}_r = (\mathbf{q}_{||}, q_{z,r})$$

Fresnel's law (wave amplitude) and Snell's law (wave vector)



$$T_1 = 1 \text{ and } R_N = 0.$$

Fresnel's law (wave amplitude) and Snell's law (wave vector)

The Fresnel coefficients at the interface between layers m and $m+1$ are

$$r_{m,m+1} = \frac{k_{z,m} - k_{z,m+1}}{k_{z,m} + k_{z,m+1}}$$

$$t_{m,m+1} = \frac{2k_{z,m}}{k_{z,m} + k_{z,m+1}},$$

$$X_m = \frac{R_m}{T_m} = e^{2ik_{z,m}z_m} \left(\frac{r_{m,m+1} + X_{m+1}e^{2ik_{z,m+1}z_m}}{1 + r_{m,m+1}X_{m+1}e^{-2ik_{z,m+1}z_m}} \right)$$

$$X_m = \frac{R_m}{T_m} = e^{-2ik_{z,m}z_m} \left(\frac{r_{m,m+1} + X_{m+1}e^{2ik_{z,m+1}z_m}}{1 + r_{m,m+1}X_{m+1}e^{2ik_{z,m+1}z_m}} \right)$$

$$R_{m+1} = \frac{1}{t_{m+1,m}} \left(T_m r_{m+1,m} e^{i(k_{z,m+1} + k_{z,m})z_m} + R_m e^{i(k_{z,m+1} - k_{z,m})z_m} \right) \quad R_{m+1} = \frac{1}{t_{m+1,m}} \left(T_m r_{m+1,m} e^{-i(k_{z,m+1} + k_{z,m})z_m} + R_m e^{-i(k_{z,m+1} - k_{z,m})z_m} \right)$$

$$T_{m+1} = \frac{1}{t_{m+1,m}} \left(T_m e^{-i(k_{z,m+1} - k_{z,m})z_m} + R_m r_{m+1,m} e^{-i(k_{z,m+1} + k_{z,m})z_m} \right) \quad T_{m+1} = \frac{1}{t_{m+1,m}} \left(T_m e^{i(k_{z,m+1} - k_{z,m})z_m} + R_m r_{m+1,m} e^{i(k_{z,m+1} + k_{z,m})z_m} \right),$$

Intensity and scattering vectors

$$\Psi_{sc,m}(R) = -\frac{e^{ik_0 R}}{R} \int \Psi_{f,m}^*(\mathbf{r}) V(\mathbf{r}) \Psi_{i,m}(\mathbf{r}) d\mathbf{r},$$

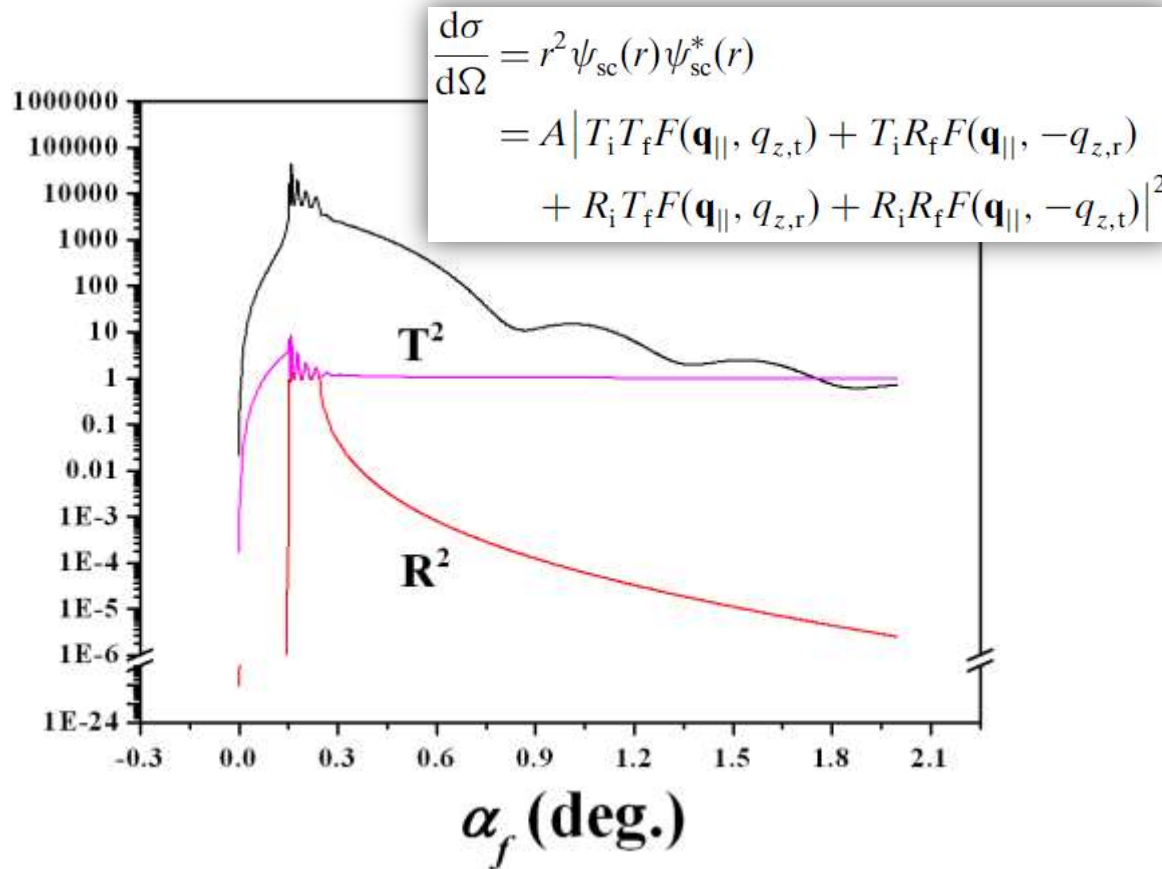
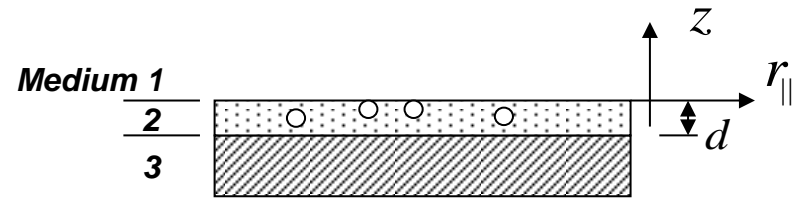
$$\Psi_{sc,m}(R) = \frac{e^{ik_0 R}}{R} \Phi_m(\mathbf{q}),$$

$$\begin{aligned} \Phi_m(\mathbf{q}) = & T_{i,m} T_{f,m} A(\mathbf{q}_{||,m}, q_{t,z,m}) + T_{i,m} R_{f,m} A(\mathbf{q}_{||,m}, -q_{r,z,m}) \\ & + R_{i,m} T_{f,m} A(\mathbf{q}_{||,m}, q_{r,z,m}) + R_{i,m} R_{f,m} A(\mathbf{q}_{||,m}, -q_{t,z,m}), \end{aligned}$$

where $A(\mathbf{q})$ is the Fourier transform of $V(\mathbf{r})$

$$I(\mathbf{q}) = \sum_m R^2 \Psi_{sc,m}(R) \Psi_{sc,m}^*(R) = \sum_m \Phi_m(\mathbf{q}) \Phi_m^*(\mathbf{q})$$

Effect of wave amplitudes



References

Probing surfaces and interfaces morphology with Grazing Incidence Small Angle X-Ray Scattering

G. Renaud *et al.*, **Surface Science Reports**, 64(8), 255-380 (2009).

Structural Analysis of Block Copolymer Thin Films with Grazing Incidence Small-Angle X-ray Scattering

Byeongdu Lee *et al.*, **Macromolecules**, 38(10), 4311-4323 (2005).

Softwares

<http://ln-www.insp.upmc.fr/axe2/Oxydes/IsGISAXS/isgisaxs.htm>

<http://www.chemie.uni-hamburg.de/pc/sfoerster/software.html>

<http://sites.google.com/site/byeongdu/software>

